Synthesis of CIGS Powder by Mechanochemical Method†

SIN-IL GU^{1,2}, HYU-SOON SHIN¹, DONG-HUN YEO^{1,*} and SAHN NAHM²

¹Advanced Materials Convergence Division, Korea Institute of Ceramic Engineering and Technology 233-5, Gasan-Dong, Guemcheon-Gu Seoul 153-801, Republic of Korea

²Department of Materials Science and Engineering, Korea University, Seoul 136-701, Republic of Korea

*Corresponding author: E-mail: ydh7@kicet.re.kr

Published online: 1 March 2014;

AJC-14828

To this day $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$ (CIGS) powder used in the manufacture of thick-film CIGS photovoltaic (PV) cells was mainly produced using the solvothermal method. However, this method has drawbacks such as it is expensive and cannot be used to mass produce the CIGS powder. Thus, there is a need to develop a more cost-effective manufacturing process for mass production of CIGS powder to develop stable thick-film CIGS PV cells. The mechanochemical method, which facilitates mass production, can be considered as an alternative for the synthesis of CIGS, but such research results have yet to be reported. In this study, the CIGS synthesis pathway using the mechanochemical method was determined, through which single-phase CIGS was synthesized. Single-phase CIGS was synthesized when the molar ratio x of Ga was below 0.15 in the raw material, $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$, but not above this ratio. Also, adding CGS to the CIGS synthesized using the mechanochemical method for a secondary synthesis resulted in increased proportion of Ga in CIGS. Based on the results, it was confirmed that CIGS powder could be synthesized in large quantities at a low cost using the simple process of milling.

Keywords: $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$ powder, Mechanochemical method, Thick film process, Solar cell.

INTRODUCTION

There has recently been an increase of research on thick-film $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$ (hereafter CIGS) photovoltaic cell as it is easy and inexpensive to increase the area¹. In order to manufacture thick-film CIGS photovoltaic cells, there is a need to develop a process that makes it possible to manufacture CIGS powder in large quantities at low costs². Thus, there has mainly been research on synthesizing CIGS powder applicable to the thick-film process using the solvothermal method^{3,4}. However, the solvothermal method has drawbacks in that it is costly and poses difficulties in mass production⁵. Thus, there is a need for another method that can be used to synthesize CIGS in an easier and more cost-effective manner.

The advantage of using the mechanochemical method is that it can synthesize powder in large quantities easily^{6,7}. Thus, various types of powder have been synthesized using this method^{8,9}. The results of studies on synthesis of CIS, which is simpler than synthesis of CIGS, for the manufacture of thick-film PV cells, have been reported by Wu *et al.*¹⁰ and Wada *et al.*¹¹. However, there have not been any results on synthesizing single-phase CIGS using the mechanochemical method.

The aim of this study was to synthesize the powder from the raw ingredients, Cu, In, Ga and Se, using the mechanochemical method and to understand the CIGS synthesis pathway by analyzing the crystal structure and changing the processes. The ultimate aim was to synthesize single-phase CIGS powder in large quantities that can be applied to thick-film PV cell using the results of the experiment.

EXPERIMENTAL

The powders of Cu (99.7 %, Sigma-Aldrich Co. Ltd., Japan), In (99.99 %, Sigma-Aldrich Co. Ltd., Japan), Ga (99.99 %, Sigma-Aldrich Co. Ltd., Japan) and Se (99.99 %, Sigma-Aldrich Co. Ltd., Japan) were used. And the solvent was used diethylamine (99.9 %, Sigma-Aldrich Co. Ltd., Japan).

In a glove box with nitrogenous atmosphere, Cu, In, Ga and Se powders, 60 g of 3Φ zirconia balls and 1.2 cc of diethylamine were loaded into a 45cc zirconia jar. The molar ratio of In and Ga was varied during the experiment and the total amount of powder was 10 g. For the zirconia jar, planetary mill (Pulverisette 7, Fritsch GmbH, Germany) was used to perform the milling process for 1 h at 300 rpm, after which the powder was dried for 24 h at 80 °C.

†Presented at The 7th International Conference on Multi-functional Materials and Applications, held on 22-24 November 2013, Anhui University of Science & Technology, Huainan, Anhui Province, P.R. China

In a glove box with nitrogenous atmosphere, the CIGS powder (7.3 g) synthesized in the primary synthesis process and the additional raw materials of Cu (0.192 g), Ga (0.248 g) and Se (0.519 g) were weighed and loaded into a 45 cc zirconia jar with 60 g of 3 Φ zirconia balls and 1.2 cc of diethylamine. For the zirconia jar, planetary mill was used to perform the milling process for 1 hr at 300 rpm, after which the powder was dried for 24 h at 80 °C.

The crystal structures of synthesized powders were measured by X-ray diffractometer (PANalytical, X'pert pro, Netherlands) and the particle morphologies were observed by field emission scanning electron microscope (LSM-6700F, Jeol, Japan).

RESULTS AND DISCUSSION

Fig. 1 shows X-ray diffraction patterns of synthesized powder with composition of (a) CuInSe₂, (b) CuIn_{0.5}Ga_{0.5}Se₂ (c) CuGaSe₂. Synthesis of CIS using the mechanochemical method produced single-phase CIS. Also, the results of the preceding experiment showed that the powder particle size could be controlled during CIS synthesis by adding a solvent¹². However, in case of CuIn_{0.5}Ga_{0.5}Se₂, in which 50 % of In was substituted with Ga, In and Se peaks were observed and CuSe₂ was produced as shown in (b). As for CuGaSe₂, where In was completely substituted with Ga, no peaks were observed, while CuSe and unknown phases were observed. Based on these results, it was determined that during synthesis of CIGS powder using the mechanochemical method, an increase of Ga led to formation of phases that were not consistent with the added metal sources. This is thought to be caused by the fact that Ga, one of the raw materials, is in liquid form at room temperature and absorbs the shock between the balls during the milling process, thereby interfering with the synthesis. This result is consistent with the result of increasing the amount of solvent added during the synthesis of CIS powder, reported by Gu *et al.*¹². Adding a small amount of solvent does not interfere with the synthesis of CIS and enhances the milling effect, whereas excess solvent absorbs shock and interferes with the synthesis of CIS. This reason also applies in this case as the Ga metal source is in its liquid phase during the reaction.

Fig. 1. X-Ray diffraction patterns of synthesized powder with composition of (a) CuInSe₂, (b) CuIn_{0.5}Ga_{0.5}Se₂ and (c) CuGaSe₂

Thus, it was determined that a low molar ratio of Ga would not interfere with the reaction when synthesizing CIGS using the mechanochemical method. Fig. 2 shows the X-ray diffraction patterns of synthesized powder with the variation of Ga ratios of material source. Single-phase CIGS was synthesized until the molar ratio of Ga was 0.15. However, when the molar ratio of Ga exceeded 0.2, there was presence of In, Se and CuSe without the synthesis of CIGS. Based on this, it was determined that adding a small amount of Ga does not interfere with the synthesis of CIGS, but adding an amount exceeding a particular ratio led to interference with the synthesis of single-phase CIGS.

Fig. 2. X-Ray diffraction patterns of synthesized powder with the variation of Ga ratios of material source

Fig. 3 showed the Ga ratios of synthesized CIGS phase with the variation of Ga ratios of material source. Fig. 3 shows the Ga ratios of synthesized CIGS phase with the variation of Ga ratios of material source. Because the metal source, Ga, is in its liquid phase at room temperature, observation is difficult through the X-ray diffraction pattern. Also, there is a need to check how much Ga has been incorporated into CIGS as it can exist as a mixture without being incorporated. The composition was checked based on the locations of the X-ray diffraction peaks using the synthesized CIGS powder and it showed that the proportion of Ga in the synthesized CIGS increased constantly with increasing proportion of Ga in the raw sources. In addition, based on the observation that the proportions of Ga in the raw sources and in the synthesized CIGS were consistent with each other, it was determined that all the Ga from the raw sources was incorporated into CIGS. From these results, it was determined that adding a small amount of Ga leads to 100 % participation in the reaction without interfering with the synthesis of CIGS.

The efficiency of CIGS PV cell varies depending on the ratio of In and Ga in CIGS. The ratio of In and Ga in a CIGS PV cell, which is considered most efficient to this day, is 7:3¹³. However, the molar ratio of Ga that can be incorporated into CIGS through one milling process is only 0.15 as confirmed in Fig. 3. Thus, Cu, Ga and Se were added to the synthesized CuIn_{0.85}Ga_{0.15}Se₂ for a secondary synthesis. Fig. 4 is the X-ray diffraction pattern of a) one time synthesis powder and (b) two time synthesis powder with the addition of CGS source.

Fig. 3. Ga ratios of synthesized CIGS phase with the variation of Ga ratios of material source

Fig. 4. X-Ray diffraction patterns of (a) one time synthesis powder and (b) two time synthesis powder with the addition of CGS source

Adding the raw sources, Cu, Ga and Se, to the $\text{CuIn}_{0.85}\text{Ga}_{0.15}\text{Se}_2$ produced after the primary synthesis resulted in the synthesis of $\text{CuIn}_{0.74}\text{Ga}_{0.26}\text{Se}_2$. Through this, it was indirectly confirmed that it was possible for Cu, Ga and Se to additionally be incorporated into the synthesized CIGS.

Fig. 5 shows the morphologies of (a) one time synthesis powder and (b) two time synthesis powder. The synthesized $\text{CuIn}_{0.85}\text{Ga}_{0.15}\text{Se}_2$ varied greatly in size, from hundreds of nm to dozens of μm . $\text{CuIn}_{0.74}\text{Ga}_{0.26}\text{Se}_2$ synthesized through a secondary synthesis process also showed similar sizes. It was determined that there were no significant differences in the size between these two powders because milling was sufficiently performed in the first hour.

Fig. 5. Morphologies of (a) one time synthesis powder and (b) two time synthesis powder

Conclusions

The results of CIGS synthesis using the mechanochemical method with Cu, In, Ga and Se as the raw sources are as follows:

- Single-phase CIGS was only synthesized when the molar ratio of Ga in the raw source, $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$, was less than 0.15 and all the Ga present in the raw source was incorporated into CIGS.
- By adding the raw sources, Cu, Ga and Se, to the synthesized $\text{CuIn}_{0.85}\text{Ga}_{0.15}\text{Se}_2$, the amount of Ga incorporated into the synthesized CIGS was increased.
- The synthesized CIGS varied greatly in size and the CIGS powder synthesized secondarily by adding the raw materials of CGS had similar size variations and shapes.

REFERENCES

1. S.J. Ahn, K.H. Kim and K.H. Yoon, *Colloids Surf. A*, **313-314**, 171 (2008).
2. S.J. Ahn, K.H. Kim, Y.G. Chun and K.H. Yoon, *Thin Solid Films*, **515**, 4036 (2007).
3. S.I. Gu, H.S. Shin, D.H. Yeo, Y.W. Hong and S. Nahm, *Curr. Appl. Phys.*, **11**, S99 (2011).
4. Y.-G. Chun, K.-H. Kim and K.-H. Yoon, *Thin Solid Films*, **480-481**, 46 (2005).
5. X. Zhang, H. Liu, W. He, J. Wang, X. Li and R.I. Boughton, *J. Cryst. Growth*, **275**, e1913 (2005).
6. S. Ohara, H. Abe, K. Sato, A. Kondo and M. Naito, *J. Eur. Ceram. Soc.*, **28**, 1815 (2008).
7. T.D. Isfahani, J. Javadpour, A. Khavandi, R. Dinnebier, H.R. Rezaie and M. Goodarzi, *Int. J. Refrac. Metals Hard Mater.*, **31**, 21 (2012).
8. T. Rojac, O. Masson, R. Guinebretiere, M. Kosec, B. Malic and J. Holc, *J. Eur. Ceram. Soc.*, **27**, 2265 (2007).
9. S.I. Gu, H.S. Shin, D.H. Yeo and S. Nahm, *Asian J. Chem.*, **24**, 4213 (2012).
10. S. Wu, Y. Xue and Z. Zhang, *J. Alloys Comp.*, **491**, 456 (2010).
11. T. Wada and H. Kinoshita, *Thin Solid Films*, **480-481**, 92 (2005).
12. S.I. Gu, H.S. Shin, D.H. Yeo, S. Nahm, *J. Ceram. Process Res.*, (Submitted).
13. I. Repins, M.A. Contreras, B. Egaas, C. DeHart, J. Scharf, C.L. Perkins, B. To and R. Noufi, *Prog. Photovolt. Res. Appl.*, **16**, 235 (2008).