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Supporting information for article:

Local structure and stacking disorder of chloro(phthalocyaninato)aluminium

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## S1. Tailor-made force field parameters

The standard Dreiding force field does not have parameters for an $\mathrm{Al}^{3+}$ ion with square-planar coordination. Hence, new force field types were added to provide a realistic description of the molecular structure of AlPcCl (see Table S1). The values for all new parameters are shown in 0.

Table S1 Added force field types used for AlPcCl .

| force field types | description |
| :--- | :--- |
| C_R1 | aromatic $C$ atom of the phthalocyanine moiety (type 1) |
| C_R2 | aromatic $C$ atom of the phthalocyanine moiety (type 2) |
| C_R3 | aromatic C atom of the phthalocyanine moiety (type 3) |
| N_R1 | inner $N$ atom of the phthalocyanine ring (type 1) |
| N_R2 | outer $N$ atom of the phthalocyanine ring |

Table S2 Values for all introduced force field types (bond lengths, bond angles and torsion angles) used for AlPcCl.

| bond length | $R_{0} / \AA$ | $k_{B} /\left(\mathrm{kcal} \cdot \mathrm{mol}^{-1} \cdot \AA^{2}\right)$ |
| :--- | :--- | :--- |
| Al3-Cl | 2.179 | 700 |
| Al3-N_R1 | 1.976 | 700 |
| Al3-N_R2 | 1.976 | 700 |
| C_R1-C_R2 | 1.407 | 700 |
| C_R1-N_R3 | 1.263 | 700 |
| C_R2-C_R2 | 1.430 | 700 |
| C_R2-C_R3 | 1.463 | 700 |
| C_R2-H_ | 1.080 | 700 |
| C_R3-C_R3 | 1.409 | 7.140 |


| N_R1-C_R1 | 1.442 | 700 |
| :--- | :--- | :--- |
| N_R2-C_R1 | 1.442 | 700 |


| bond angle | $\vartheta_{0} /{ }^{\circ}$ | $k_{\theta} /\left(\mathrm{kcal} \cdot \mathrm{mol}^{-1}\right)$ |
| :---: | :---: | :---: |
| Al3-N_R1-C_R1 | 127.827 | 200 |
| Al3-N_R2-C_R1 | 127.827 | 200 |
| C_R1-C_R2-C_R2 | 108.767 | 200 |
| C_R1-C_R2-C_R3 | 123.041 | 200 |
| C_R1-N_R3-C_R1 | 127.372 | 200 |
| C_R1-N_R3-C_R2 | 125.577 | 200 |
| C_R2-C_R2-C_R2 | 115.403 | 200 |
| C_R2-C_R2-C_R3 | 123.759 | 200 |
| C_R2-C_R2-H_ | 122.496 | 200 |
| C_R2-C_R3-C_R3 | 117.102 | 200 |
| C_R2-C_R3-H_ | 122.496 | 100 |
| C_R3-C_R3-C_R3 | 122.168 | 200 |
| $\mathrm{Cl}-\mathrm{Al} 3-\mathrm{N}, \mathrm{R} 1$ | 102.308 | 200 |
| $\mathrm{Cl}-\mathrm{Al} 3-\mathrm{N}, \mathrm{R} 2$ | 102.308 | 200 |
| N_R1-Al3-N_R1 | 156.065 | 200 |
| N_R1-Al3-N_R2 | 87.567 | 200 |
| N_R1-C_R1-C_R2 | 107.937 | 200 |
| N_R1-C_R1-N_R3 | 118.801 | 200 |
| N_R2-Al3-N_R2 | 156.065 | 200 |
| N_R2-C_R1-C_R2 | 107.937 | 200 |
| N_R2-C_R1-N_R3 | 118.801 | 200 |
| N_R3-C_R1-C_R2 | 126.528 | 200 |


| X-C_R3-H_ | 120.800 | 100 |
| :--- | :---: | :---: |
| X-N_R1-X | 120.000 | 100 |
| X-N_R2-X | 120.000 | 100 |
| X-N_R3-X | 120.000 | 100 |


| torsion angle | B | $n$ | $d$ |
| :--- | :--- | :--- | :--- |
| Cl-Al3-X-X | 25 | 2 | -1 |
| X-Al3-N_R1-X | 25 | 2 | 1 |
| X-Al3-N_R2-X | 25 | 2 | 1 |
| X-C_R1-C_R2-X | 25 | 2 | 1 |
| X-C_R1-N_R1-X | 25 | 2 | 1 |
| X-C_R1-N_R2-X | 25 | 2 | 1 |
| X-C_R1-N_R3-X | 25 | 2 | 1 |
| X-C_R2-C_R3-X | 25 | 2 | 1 |
| X-C_R2-C_R2-X | 25 | 2 | 1 |
| $X-C \_R 3-C \_R 3-X ~$ | 25 | 2 | 1 |

## S2. Validation of the force field

The modified force field was tested on the ordered crystal structures ZnPcCl (Mossoyan-Deneux et al., 1985), InPcl (Janczak \& Kubiak et al., 1999), TiPcO (Oka et al., 1992) and GaPcCl (Wynne, 1984). A Superposition of these crystal structures and the force field optimised structures are shown in Figure S1 and Figure S2.


Figure S1 Superposition of crystal structures determined by single-crystal X-ray analysis and force field optimised crystal structures. a) Experimental crystal structure of ZnPcCl (Mossoyan-Deneux et al., 1985) (green), crystal structure optimised by force field (yellow). b) Experimental crystal structure of InPcI (Janczak \& Kubiak et al., 1999) (green), crystal structure optimised by force field (yellow).


Figure S2 Superposition of crystal structures determined by single-crystal X-ray analysis and force field optimised crystal structures. a) Experimental crystal structure of TiPcO (Oka et al., 1992) (green), crystal structure optimised by force field (yellow). b) Experimental crystal structure of GaPcCl (Wynne, 1984) (green), crystal structure optimised by force field (yellow).

For comparison of the experimental and the energy-minimised crystal structures the root mean square Cartesian displacement (excluding H atoms) is used (van de Streek \& Neumann, 2010; van de Streek \& Neumann, 2014). The upper r.m.s.c.d. limit for a correct structure must be increased from 0.25 Å

Table S3 R.m.s. Cartesian displacements of the force field minimised structures versus the experimental crystal structures.

| CSD-refcode | Compound | r.m.s.c.d. value $/ \AA$ |  |
| :--- | :--- | :---: | :---: |
|  |  | fixed unit cell | free unit cell |
| DAMTEQ | ZnPcCl | 0.1865 | 0.2174 |
| BIJQOA | InPcl | 0.1366 | 0.1404 |
| BITSAYO2 | TiPcO | 0.4791 | 0.4576 |
| CUWNUD | GaPcCl | 0.1352 | 0.4089 |

The experimental crystal structures of ZnPcCl and InPcl were reproduced very well and the r.m.s.c.d. values are small. After optimisation of the crystal structure of TiPcO, the packing in principle is reproduced well, but the molecules are slightly shifted resulting in an increased r.m.s.c.d. value. In the structure of GaPcCl the molecules of the structure optimised by the force field are inclined to their original position. Therefore the r.m.s.c.d. value is large. But in principal the arrangement of the molecules agree with the experimental one.

## S3. Calculation of stacking probabilities

For the stacking sequences, an additive combination of energies is assumed, i.e. the energy of a given double-layer depends almost exclusively only on the arrangement of the neighbouring layers, whereas interactions with more distant double-layers do not play a major role. For a given length of the local stacking motif, we use a combinatorial approach to get the total number of possible local stacking motifs (Teteruk et al, 2014).

For a statistical examination of the energies of the stacking motifs, the Boltzmann distribution is used.

$$
\frac{N_{i}}{N_{j}}=\frac{g_{i}}{g_{j}} \mathrm{e}^{\frac{E_{i}-E_{j}}{k T}}
$$

Using the energies of the stacking motifs, first the relative occupancy values $N_{i}$ and afterwards stacking probabilities $p_{i}$ for each local stacking motif was calculated.

$$
p_{i}=\frac{N_{i}}{\sum_{i=1}^{n} N_{i}}
$$

As the basis we took the DFT-D optimised models ' $11111^{\prime}$ with $Z=2$, ' 1212 ' with $Z=4$ and ' 1221 ' with $Z=8$. On a given double-layer the obtained stacking probabilities describe the chance to find a specific neighbouring double-layer within a theoretical infinite disordered crystal. The resulting stacking probabilities are given in Table S4.

Table S4 Probabilities for local stacking motifs. The Energies are given in $\mathrm{kJ} \cdot \mathrm{mol}^{-1}$ per doublelayer.

| Local stacking motif | $E_{\text {rel }}$ of the local stacking motif <br>  <br>  <br>  <br> $\mathrm{kJ} \cdot \mathrm{mol}^{-1}$ | Probability of the local stacking motif |
| :--- | :--- | :--- |
| 111 | 3.40 | 0.06 |
| 112 | 2.13 | 0.10 |
| 121 | 0.00 | 0.24 |
| 122 | 2.13 | 0.10 |
| 211 | 2.13 | 0.10 |
| 212 | 0.00 | 0.24 |
| 221 | 2.13 | 0.10 |
| 222 | 3.40 | 0.06 |

S4. Calculation of the interaction between the $1^{\text {st }}$ and $3^{\text {rd }}$ double-layers
The interaction energy between the $1^{\text {st }}$ and the $3^{\text {rd }}$ double-layers connected by a shift of $\boldsymbol{t}_{1}+\boldsymbol{t}_{2}$ were calculated using the model '12'. The $\boldsymbol{c}$ axis was quadrupled ( $\boldsymbol{c}$ ' $=4 \boldsymbol{c}$ ), so that the unit cell include 8 double-layers. All molecules, except those in $1^{\text {st }}$ and $3^{\text {rd }}$ double-layers, were removed. The energy was calculated with force field methods without optimising the cell parameters and coordinates of all atoms. Afterwards the $3^{\text {rd }}$ double-layer were shifted in such a way, that it corresponds to a shift of $2 \cdot \boldsymbol{t}_{1}$ (related to the $1^{\text {st }}$ double-layer). The energy was calculated in the same way as before. This is the interaction energy between the $1^{\text {st }}$ and the $3^{\text {rd }}$ double-layers connected by a shift of $2 \cdot \boldsymbol{t}_{1}$.

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