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Single-crystal quality data from polycrystalline samples: finding the needle in the haystack

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# Single-crystal quality data from polycrystalline samples: finding the needle in the haystack 

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## S1. Sample preparation

## Co-crystal of $\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}$ :

1.86 g of hexafluorobenzene $\left(\mathrm{C}_{6} \mathrm{~F}_{6}, 0.01 \mathrm{~mol}\right.$, Sigma Aldrich Ltd., H8706, M.W. $=186.05,99 \%$ purity) and 0.67 g of pyrrole $\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}, 0.01 \mathrm{~mol}\right.$, Sigma Aldrich Ltd. 131709 , M.W. $=67.09,98 \%$ purity) were added to a glass sample vial giving a clear, homogeneous equimolar liquid mixture. Subsequently, a second sample was prepared for DSC in the correct 3:4 ratio to match the co-crystal structure.

## Co-crystal of $\mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}$ :

1.86 g of hexafluorobenzene $\left(\mathrm{C}_{6} \mathrm{~F}_{6}, 0.01 \mathrm{~mol}\right.$, Sigma Aldrich Ltd., H8706, M.W. $=186.05,99 \%$ purity) and 0.79 g of pyridine $\left(\mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}, 0.01 \mathrm{~mol}\right.$, Sigma Aldrich Ltd., 360570 , M.W. $=79.10, \geq 99 \%$ purity) were added to a glass sample vial giving a clear, homogeneous liquid mixture.

## Co-crystal of $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ and $p-\mathrm{C}_{8} \mathrm{H}_{10}$ :

1.11 g of pentafluorobenzene $\left(\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}, 0.01 \mathrm{~mol}\right.$, Fluorochem, 002836 , Batch FCB 114752 , M.W. $=$ $168.06,99 \%$ purity $)$ and 1.23 g of $p$-xylene $\left(\mathrm{C}_{8} \mathrm{H}_{10}, 0.01 \mathrm{~mol}\right.$, Sigma-Aldrich, $95682-2.5 \mathrm{~L}$, $\mathrm{MW}=106.17, \mathrm{GC}$ grade $\geq 99 \%$ ) were added to a glass sample vial giving a clear, homogeneous liquid mixture.

## S2. Single-crystal X-ray diffraction (SXD) experiments

X-ray diffraction data on crystals of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ were obtained using an Agilent Oxford Diffraction SuperNova diffractometer equipped with microfocus X -ray sources, and upgraded with a Rigaku HyPix Arc-100 detector (Fig. S1). X-ray diffraction data on $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ were obtained on the Agilent Oxford Diffraction SuperNova diffractometer equipped using the older Atlas CCD detector, shortly before the installation of the new HyPix one. All X-ray diffraction data were obtained with $\mathrm{Cu} \mathrm{K} \alpha$ radiation $(\lambda=1.54184 \AA)$.

## S2.1. Crystal growing from equimolar mixture of $\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}$

The flame-sealed capillary containing $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ was mounted on the diffractometer and cooled rapidly in situ to 230 K resulting in the formation of numerous white crystals. The temperature was then increased to 260 K , (just below the melting point at 266 K ) in order to encourage annealing of single crystals present. The sample was held at 260 K for 2 hours before cooling to 254 K for data acquisition. Post structure solution, it was found that the co-crystal from an equimolar mixture of $\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}$ did not crystallise in a 1:1 molar ratio, but rather in a 3:4 ratio, which explained the difficulties encountered when trying to grow single crystals of this material. The use of the wrong ratio resulted in additional unwanted diffraction signal from the excess $\mathrm{C}_{6} \mathrm{~F}_{6}$ present.

## S2.2. Crystal growing from equimolar mixture of $\mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}$

After mounting on the diffractometer, the capillary containing $\mathrm{C}_{6} \mathrm{~F}_{6}$ : $\mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ was cooled rapidly in situ to 200 K resulting in the formation of numerous white crystals. The temperature was then raised to 230 K , and crystal growth was encouraged by manual slow translation of the capillary sample through the cold $\mathrm{N}_{2}$ gas stream. Data was initially acquired at 200 K , and then subsequently a full-sphere of data was measured at 150 K .

## S2.3. Crystal growing from equimolar mixture of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ and $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$

After mounting on the diffractometer, the capillary containing $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ was cooled rapidly in situ to 200 K resulting in the formation of numerous clear/white crystals. The temperature was then raised to just below the melt at about 255 K , and crystal growth was encouraged by manual slow translation of the capillary sample through the cold $\mathrm{N}_{2}$ gas stream. A full-sphere of data was initially acquired at 230 K , and then subsequently at $160 \mathrm{~K}, 200 \mathrm{~K}$, and finally at 120 K . It is noteworthy that the system underwent a phase transition between 160 K and 120 K (later confirmed by DSC), which saw a doubling of the unit cell volume on cooling.

## S2.4. Data acquisition, reduction, and structure solution and refinement

Full spheres of X-ray diffraction data were collected to $0.84 \AA$ resolution using $0.5^{\circ} \omega$ scan frames; 6124 frames at 2 s per frame for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ and 6496 frames at 0.75 s per frame for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$. This corresponds to a total collection time of under 4 hrs for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ and about 1 hr 45 min for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$. The Cryojet5® used for the measurements in this study is the original prototype device developed by Oxford Instruments and the Pt-resistance sensor is located in the copper-block heat exchanger and not in the nozzle of the instrument close to the sample. Thus, the temperatures quoted in these SXD experiments should be treated as nominal (despite stability to much better than 0.1 K ), especially since the sample mass was considerably larger than for a normal singlecrystal experiment. Automatic data reduction was performed using the CrysAlis PRO software package from Rigaku Oxford Diffraction, version 1.171.42.60a for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ and $p$ $\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$, and the later version 1.171.42.72a for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$.

The crystal structures of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ and $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K were solved using the program SHELXT ${ }^{1}$ and refined by least-squares using the refinement program SHELXL ${ }^{2}$ within the Olex2 program suite. ${ }^{3}$ The positions of all atoms were refined freely, with isotropic displacement parameters used for H atoms. Crystal structures are illustrated with the program Mercury ${ }^{4}$ from CCDC with thermal ellipsoids (excluding H atoms) shown at $50 \%$ probability, H atoms are shown as fixed radius spheres. CIF files have been uploaded to the Cambridge Crystallographic Data Centre with deposition numbers 2263896 and 2263897.

[^0]As previously stated, data sets were measured on $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 4 different temperatures: $200 \mathrm{~K}, 160 \mathrm{~K}, 230 \mathrm{~K}$, and then 120 K . At 200 K , peak peaking software and automatic indexing identified a triclinic unit cell from around $39 \%$ of the visible spots. The high temperature phase I was processed in space group $P \overline{1}$ with $a=6.4137(5) \AA, b=7.2398(5) \AA, b=7.3932(5) \AA$, $\alpha=107.364(6)^{\circ}, \beta=94.389(6)^{\circ}, \gamma=99.370(6)^{\circ}$, and $V=320.41(4) \AA^{3}$. Structure solution using SHELXT ${ }^{1}$ suggested that phase I exhibited orientational disorder of the $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ rings. Cooling to low temperature revealed the existence of phase II. This phase transition was not known in advance but its existence was suspected due to the disorder observed at higher temperatures. Due to instrument availability, SXD measurements were made prior to DSC and PXRD data collection on this sample. At 120 K , a triclinic cell was identified from $45 \%$ of the visible spots with $a=8.567 \AA, b=8.779 \AA$, $c=9.371 \AA, \alpha=79.94^{\circ}, \beta=69.33^{\circ}, \gamma=70.14^{\circ}$, and $V=619.0 \AA^{3}$, for which the volume is roughly double that observed at higher temperatures. However, this unit cell does not easily relate to that of phase I. Subsequently, a larger I-centred triclinic unit cell was chosen such that the molecular column axis in both phases is in the same direction along $c$. The refined parameters for this cell are:
$a=10.2296(5) \AA, b=8.7819(5) \AA, c=14.6776(7) \AA, \alpha=106.811(5)^{\circ}, \beta=80.111(4)^{\circ}, \gamma=97.168(4)^{\circ}$, and $V=1239.50(11) \AA^{3}$.

The structures of phases I and II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ were refined by least-squares using the refinement program SHELXL ${ }^{2}$ within the Olex2 program suite. ${ }^{3}$ The positions of all atoms were refined freely, with isotropic displacement parameters used for H atoms. In phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$, the centre of mass of both rings sit on a point of inversion within space group $P \overline{1}$. Refinement of the fluorine occupancy for $\mathrm{F}(1), \mathrm{F}(2)$, and $\mathrm{F}(3)$ indicated that the single H atom in $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ was unevenly distributed across all three possible positions (see Fig. S17). In the least-squares refinement, the total occupancy of the fluorine sites was tightly restrained to 5 per molecule such that the molecular formula is exactly $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$. It is not possible to locate the single H atom of $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ in phase I and so its electron density was constrained to be at the same position as the F atoms. In phase I, the methyl group of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ was assumed to be disordered and is modelled with 6 positions for the three H atoms. In phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$, there are two crystallographicallydistinct $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ molecules, both of which sit on a point of inversion within space group $\bar{I} \overline{1}$. The methyl group was modelled as ordered using the AFIX 137 instruction. For $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$, the site occupancy of all 6 possible F positions were refined: one fluorine showed a low occupancy while the occupancy of sites $\mathrm{F}(1)$ and $\mathrm{F}(4)$ refined to nearly $100 \%$. The latter were therefore fixed as fully occupied. In this phase, the H atom is mainly located on one site but there is still some residual disorder, possible due to the fact that the measurement was unwittingly made close to the phase-transition temperature. The position of $\mathrm{H}(2)$ was refined to be co-axial with the residual fluorine density on this site, but residual H scattering was distributed on to the $\mathrm{F}(3), \mathrm{F}(5)$ and $\mathrm{F}(6)$ sites (see Fig. S18). As with phase I, the total occupancy of the fluorine sites was heavily restrained to maintain the correct molecular formula of $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$. CIF files have been uploaded to the Cambridge Crystallographic Data Centre with deposition numbers 2287264-2287267.

## S3. Powder X-ray diffraction (PXRD) experiments

A small amount of the liquid mixture of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ was pipetted into the end of a 1.0 mm X ray capillary, shaken to the closed end, and then carefully flame-sealed to prevent sample loss, particularly of the more volatile component. The flame-sealed end was checked visually for integrity and the sample was used immediately in the PXRD experiment. Variable temperature powder X-ray diffraction (VT-PXRD) measurements were performed using a Stoe Stadi-P diffractometer equipped with a Cu anode, $\mathrm{Ge}<111>$ monochromator, a restricted height collimator to limit axial divergence, a Dectris Mythen 1K detector, and an Oxford Instruments CryojetHT (90-500 K) with an in-house modified sample setup to discourage the formation of ice on the goniometer head at low temperature.

The sample was flash-frozen by placing the capillary directly into the cold jet $\mathrm{N}_{2}$ gas of the CryojetHT precooled to 100 K to encourage formation of a crystalline powder. The temperature was raised from 100 K to room temperature in 10 K increments. At each temperature, the detector was scanned in $2 \theta$ from $2^{\circ}$ to $60^{\circ}$ in steps of $0.5^{\circ}$ at 10 s per step, a complete scan lasting approx. 30 min ; each 10 K temperature change took approx. $7-10 \mathrm{~min}$ and the sample was kept at the set temperature for 5 min before starting the next scan.

The primary purpose of our VT-PXRD analysis is the identification of phase transitions and the determination of lattice parameters and unit cell volume as a function of temperature as data collected for this purpose are not of sufficient statistical quality for structure analysis using the Rietveld method. Le Bail whole pattern fitting ${ }^{5}$ using the program Rietica ${ }^{6}$ (version 1.7.7) was used to refine the cell parameters from the data shown in Fig. S14. The results are tabulated in Table S5. It is noteworthy that large errors on the lattice parameters are likely to be obtained when the crystal system is triclinic, when a second impurity phase is present, and when the sample is not an ideal powder due to large crystallites as observed here; consequently, on several occasions inaccurate lattice parameters were obtained due to the presence of false minima and these fits had to be repeated. The variation in molecular volume (obtained by dividing the unit-cell volume by the number of molecules per cell, $Z$ ) with temperature is plotted in Fig. S15.

## S4. Differential scanning calorimetry (DSC) experiments

37.4 mg of $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ (prepared as 0.01 mol to $0.013 \mathrm{~mol}, 3: 4$ ratio) was loaded into an Al pan, covered with an Al lid, and crimped. The sample pan was then loaded into a PerkinElmer DSC8000 calorimeter at $+20^{\circ} \mathrm{C}$. A helium purge gas was used $\left(40 \mathrm{~mL} \mathrm{~min}{ }^{-1}\right)$. The sample was initially warmed to $30^{\circ} \mathrm{C}$ (heating rate $10^{\circ} \mathrm{C} \mathrm{min}^{-1}$ ) and held at this temperature for 1 min . It was then cooled to $-180^{\circ} \mathrm{C}$ at $10^{\circ} \mathrm{C} \mathrm{min}^{-1}$, held at low temperature for 1 min , and then heated back to $30^{\circ} \mathrm{C}$. The data is shown in Fig. S10. Subsequent data analysis to determine both peak maxima and peak area used the Pyris Thermal Analysis software (version 11.1.1.0492) from PerkinElmer. The data showed a sharp single exothermic peak at 240.2 K on cooling due to freezing of the liquid and a broader endothermic peak at 266.4 K on the return heating cycle due to melting of the solid. The enthalpy of fusion is

[^1]$48 \mathrm{~kJ} \mathrm{~mol}^{-1}$. It is noteworthy that an additional small exothermic peak observed at 233.3 K on cooling and the equivalent endothermic peak observed at 246.8 K on heating are attributed to a slight excess of $\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$ in the mixture from a handling loss of the more highly volatile $\mathrm{C}_{6} \mathrm{~F}_{6}$.

As there was no suggestion of a phase transition in the SXD experiments on $\mathrm{C}_{6} \mathrm{~F}_{6} \cdot \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$, no DSC data was obtained on this system. DSC data was obtained on $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ using the same protocol as for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}$. The data is shown in Fig. S16. The mass of sample used was 18.930 mg with a similar scan to that used for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$, but with the final temperature limited to $20^{\circ} \mathrm{C}$. The DSC shows a sharp exothermic peak at 248.7 K on cooling due to freezing of the liquid and a broader endothermic peak at 256.5 K on the return heating cycle due to melting of the solid. The enthalpy of fusion is $20 \mathrm{~kJ} \mathrm{~mol}^{-1}$. In addition, and in agreement with VT-PXRD, a solid-state phase transition was observed at low temperature. The enthalpy change on heating was $0.3 \mathrm{~kJ} \mathrm{~mol}^{-1}$. On cooling, the observed transition temperature was at 125.8 K while on heating the transition was at 128.7 K .

## S5. Hirshfeld surface calculations

Hirshfeld and electron density surfaces were calculated in the CrystalExplorer ${ }^{7}$ software by selecting the desired fragment and using the "Generate Surface" function. The software can generate various surfaces including Hirshfeld and electron density. The surface is visualised in terms of the normalized contact distance ( $d_{\text {norm }}$ ) which is based on both contact distances between nearest atoms present inside $\left(d_{\mathrm{i}}\right)$ and outside $\left(d_{\mathrm{e}}\right)$ the chosen fragment. For the surfaces produced for Figs. S9 and S13, it was necessary to set the pixel resolution to the highest setting available. Surfaces were not generated for on $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ due to the disorder seen in both phases.

[^2]Table S1a Crystal data and structure refinement for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at 150 K .

Identification code
Empirical formula
Formula weight
Temperature / K
Crystal system
Space group
$a / \AA$
b/Å
c / Å
$\alpha /{ }^{\circ}$
$\beta /{ }^{\circ}$
$\gamma /{ }^{\circ}$
Volume / A ${ }^{3}$
Z
$\rho_{\text {calc }} / \mathrm{g} \mathrm{cm}^{-3}$
$\mu / \mathrm{mm}^{-1}$
F(000)
Crystal size / mm ${ }^{3}$
Radiation
$2 \theta$ range for data collection $/{ }^{\circ}$
Index ranges
Reflections collected
Independent reflections
Data/restraints/parameters
Goodness-of-fit on $F^{2}$
Final $R$ indexes $[I \geq 2 \sigma(I)]$
Final $R$ indexes [all data]
Largest diff. peak/hole / e $\AA^{-3}$
exp_197
$\mathrm{C}_{34} \mathrm{H}_{20} \mathrm{~F}_{18} \mathrm{~N}_{4}$
826.54
150.0 monoclinic
$P 2{ }_{1} / n$
12.6847(3)
5.8648(2)
23.2236(5)

90
93.284(2)

90
1724.84(8)

2
1.591
1.476
828.0
$0.943 \times 0.436 \times 0.415$
$\mathrm{Cu} K \alpha(\lambda=1.54184 \AA$ Å)
7.626 to 156.186
$-15 \leq h \leq 15,-7 \leq k \leq 6,-29 \leq l \leq 29$
34525
$3554\left[R_{\text {int }}=0.1005, R_{\text {sigma }}=0.0317\right]$
3554/0/293
1.051
$R_{1}=0.0605, w R_{2}=0.1594$
$R_{1}=0.0701, w R_{2}=0.1669$
0.31/-0.26

Table S1b Fractional atomic coordinates and equivalent isotropic displacement parameters for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at $150 \mathrm{~K} . U_{\text {eq }}$ is defined as $1 / 3$ of the trace of the orthogonalised $U_{\mathrm{ij}}$ tensor.

| Atom | $\boldsymbol{x}$ | $y$ | $z$ | $U(\mathrm{eq}) / \AA^{\text {® }}$ |
| :---: | :---: | :---: | :---: | :---: |
| F(1) | 0.3034(13) | 0.1948(4) | 0.2653(8) | 0.0879(6) |
| F(2) | 0.4205(16) | -0.0817(3) | 0.2000(8) | 0.0877(6) |
| F(3) | 0.6211(16) | 0.0365(4) | 0.1796(7) | 0.0907(6) |
| F(4) | 0.7039(13) | 0.4295(4) | 0.2230(7) | 0.0834(6) |
| F(5) | 0.5862(14) | 0.7063(3) | 0.2863(7) | 0.0766(5) |
| F(6) | 0.3876(14) | 0.5882(3) | 0.3087(7) | 0.0818(5) |
| C(1) | 0.4018(19) | 0.2518(5) | 0.2547(10) | 0.0575(6) |
| C(2) | 0.4617(2) | 0.1128(4) | 0.2220(10) | 0.0596(6) |
| C(3) | 0.5634(2) | 0.1709(5) | 0.2112(9) | 0.0590(6) |
| C(4) | 0.6054(19) | 0.3712(5) | $0.2332(10)$ | 0.0558(6) |
| C(5) | 0.5451(2) | 0.5104(4) | 0.2655(10) | 0.0540(6) |
| C(6) | 0.4447(2) | 0.4516(4) | 0.2764(10) | 0.0555(6) |
| F(7) | 0.3912(11) | 0.1115(2) | -0.0254(7) | 0.0648(4) |
| F(8) | 0.4674(12) | 0.1905(3) | 0.0840(6) | 0.0651(4) |
| F(9) | 0.5755(13) | 0.5803(3) | 0.1098(6) | 0.0706(5) |
| C(7) | 0.4451(17) | 0.3034(4) | -0.0128(10) | 0.0487(5) |
| C(8) | 0.4833(17) | 0.3438(4) | 0.0427(10) | 0.0491(5) |
| C(9) | 0.5383(17) | 0.5412(4) | 0.0556(9) | 0.0498(5) |
| N(1) | 0.4935(18) | 0.2384(4) | 0.4138(9) | 0.0613(6) |
| C(10) | 0.5463(2) | 0.0807(5) | $0.3854(11)$ | 0.0643(7) |
| $\mathrm{C}(11)$ | 0.6423(3) | 0.1660(6) | $0.3762(11)$ | 0.0722(9) |
| C(12) | 0.6476(2) | 0.3863(6) | 0.4000(11) | 0.0657(7) |
| $\mathrm{C}(13)$ | 0.5544(2) | 0.4251(5) | 0.4227(11) | 0.0601(6) |
| N(2) | 0.2020(17) | 0.4650(5) | $0.0143(11)$ | 0.0654(6) |
| C(14) | 0.2079(2) | 0.3510(5) | 0.0647(15) | 0.0702(8) |
| C(15) | 0.2506(2) | 0.4895(7) | 0.1052(14) | 0.0738(9) |
| C(16) | 0.2730(2) | 0.6947(6) | $0.0791(15)$ | 0.0728(8) |
| $\mathrm{C}(17)$ | 0.2425(2) | 0.6757(5) | $0.0232(14)$ | 0.0663(7) |
| H(1) | 0.430(2) | 0.228(5) | 0.4200(12) | 0.0600(8) |
| H(10) | 0.512(3) | -0.058(7) | $0.3738(18)$ | 0.1060(13) |
| H(11) | 0.689(3) | 0.099(6) | $0.3596(16)$ | 0.0890(11) |
| H(12) | 0.703(3) | 0.478(6) | 0.3996(13) | 0.0720(9) |
| H(13) | 0.529(2) | 0.549(6) | 0.4422(14) | 0.0730(9) |
| H(2) | 0.176(3) | 0.402(6) | -0.0152(16) | 0.0860(11) |
| H(14) | 0.180(3) | 0.204(7) | 0.0670(16) | 0.0970(11) |
| H(15) | 0.267(3) | 0.456(7) | 0.139(2) | 0.1100(13) |
| H(16) | 0.310(3) | 0.823(8) | 0.0934(19) | 0.1220(14) |
| H(17) | 0.249(3) | 0.772(7) | -0.0048(16) | 0.0940(11) |

Table S1c Anisotropic displacement parameters for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at 150 K . The anisotropic displacement factor exponent has the form: $-2 \pi^{2}\left[h^{2} a^{* 2} U_{11}+2 h k a^{*} b^{*} U_{12}+\ldots\right]$.

| Atom | $\boldsymbol{U}_{11} / \AA^{2}$ | $\boldsymbol{U}_{22} / \AA^{2}$ | $\boldsymbol{U}_{33} / \AA^{2}$ | $\boldsymbol{U}_{23} / \AA^{2}$ | $\boldsymbol{U}_{13} / \AA^{2}$ | $\boldsymbol{U}_{12} / \AA^{2}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{~F}(1)$ | $0.0673(9)$ | $0.1071(15)$ | $0.0883(12)$ | $0.0168(11)$ | $-0.0048(8)$ | $-0.0165(9)$ |
| $\mathrm{F}(2)$ | $0.1241(15)$ | $0.0517(10)$ | $0.0825(11)$ | $-0.0069(8)$ | $-0.0372(10)$ | $-0.0105(9)$ |
| $\mathrm{F}(3)$ | $0.1123(13)$ | $0.0970(14)$ | $0.0617(9)$ | $-0.0190(9)$ | $-0.0047(9)$ | $0.0431(11)$ |
| $\mathrm{F}(4)$ | $0.0669(9)$ | $0.1069(15)$ | $0.0765(11)$ | $0.0189(10)$ | $0.0048(8)$ | $-0.0073(9)$ |
| $\mathrm{F}(5)$ | $0.1000(12)$ | $0.0478(9)$ | $0.0790(10)$ | $0.0005(7)$ | $-0.0211(9)$ | $-0.0111(8)$ |
| $\mathrm{F}(6)$ | $0.0872(11)$ | $0.0889(13)$ | $0.0686(10)$ | $-0.0156(9)$ | $-0.0013(8)$ | $0.0317(9)$ |
| $\mathrm{C}(1)$ | $0.0590(13)$ | $0.0613(16)$ | $0.0506(12)$ | $0.0112(11)$ | $-0.0093(10)$ | $-0.0036(11)$ |
| $\mathrm{C}(2)$ | $0.0855(17)$ | $0.0407(13)$ | $0.0496(12)$ | $0.0021(10)$ | $-0.0223(11)$ | $-0.0041(12)$ |
| $\mathrm{C}(3)$ | $0.0771(16)$ | $0.0597(16)$ | $0.0391(11)$ | $-0.0011(10)$ | $-0.0059(10)$ | $0.0187(12)$ |
| $\mathrm{C}(4)$ | $0.0613(13)$ | $0.0613(16)$ | $0.0442(11)$ | $0.0129(10)$ | $-0.0025(9)$ | $-0.0006(11)$ |
| $\mathrm{C}(5)$ | $0.0704(14)$ | $0.0384(12)$ | $0.0514(12)$ | $0.0045(9)$ | $-0.0126(10)$ | $-0.0004(11)$ |
| $\mathrm{C}(6)$ | $0.0684(14)$ | $0.0524(15)$ | $0.0446(11)$ | $0.0008(10)$ | $-0.0054(10)$ | $0.0134(11)$ |
| $\mathrm{F}(7)$ | $0.0651(8)$ | $0.0451(8)$ | $0.0834(10)$ | $-0.0050(7)$ | $-0.0029(7)$ | $-0.0062(6)$ |
| $\mathrm{F}(8)$ | $0.0731(9)$ | $0.0592(9)$ | $0.0635(8)$ | $0.0242(7)$ | $0.0079(7)$ | $0.0061(7)$ |
| $\mathrm{F}(9)$ | $0.0846(10)$ | $0.0779(11)$ | $0.0475(7)$ | $-0.0037(7)$ | $-0.0126(7)$ | $0.0014(8)$ |
| $\mathrm{C}(7)$ | $0.0480(11)$ | $0.0389(12)$ | $0.0587(12)$ | $-0.0005(9)$ | $-0.0007(9)$ | $0.0013(9)$ |
| $\mathrm{C}(8)$ | $0.0508(11)$ | $0.0464(13)$ | $0.0502(11)$ | $0.0111(9)$ | $0.0048(9)$ | $0.0060(9)$ |
| $\mathrm{C}(9)$ | $0.0516(11)$ | $0.0537(14)$ | $0.0434(10)$ | $-0.0001(9)$ | $-0.0035(8)$ | $0.0069(10)$ |
| $\mathrm{N}(1)$ | $0.0559(12)$ | $0.0640(15)$ | $0.0638(13)$ | $0.0071(10)$ | $0.0031(9)$ | $-0.0026(10)$ |
| $\mathrm{C}(10)$ | $0.0847(18)$ | $0.0541(16)$ | $0.0527(13)$ | $0.0018(11)$ | $-0.0084(12)$ | $0.0021(14)$ |
| $\mathrm{C}(11)$ | $0.0751(17)$ | $0.0970(2)$ | $0.0439(13)$ | $-0.0016(13)$ | $0.0008(12)$ | $0.0295(17)$ |
| $\mathrm{C}(12)$ | $0.0632(14)$ | $0.0800(2)$ | $0.0527(13)$ | $0.0112(13)$ | $-0.0062(11)$ | $-0.0175(14)$ |
| $\mathrm{C}(13)$ | $0.0735(16)$ | $0.0516(16)$ | $0.0547(13)$ | $0.0004(11)$ | $-0.0003(11)$ | $0.0019(12)$ |
| $\mathrm{N}(2)$ | $0.0577(11)$ | $0.0783(17)$ | $0.0594(13)$ | $-0.0208(12)$ | $-0.0026(9)$ | $0.0002(11)$ |
| $\mathrm{C}(14)$ | $0.0654(15)$ | $0.0438(16)$ | $0.1030(2)$ | $0.0052(15)$ | $0.0204(15)$ | $0.0041(12)$ |
| $\mathrm{C}(15)$ | $0.0756(17)$ | $0.0890(2)$ | $0.0567(15)$ | $0.0052(15)$ | $-0.0013(13)$ | $0.0313(16)$ |
| $\mathrm{C}(16)$ | $0.0574(14)$ | $0.0611(19)$ | $0.0980(2)$ | $-0.0273(16)$ | $-0.0111(14)$ | $0.0048(12)$ |
| $\mathrm{C}(17)$ | $0.0567(13)$ | $0.0612(18)$ | $0.0816(19)$ | $0.0202(15)$ | $0.0092(13)$ | $0.0033(12)$ |

Table S1d Selected bond lengths for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at 150 K .

| Atom - Atom | Length / $\AA$ | Atom - Atom | Length / $\AA$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)$ | $1.329(3)$ | $\mathrm{C}(7)-\mathrm{C}(8)$ | $1.371(3)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)$ | $1.343(3)$ | $\mathrm{C}(7)-\mathrm{C}(9)^{1}$ | $1.372(3)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)$ | $1.326(3)$ | $\mathrm{C}(8)-\mathrm{C}(9)$ | $1.376(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)$ | $1.330(3)$ | $\mathrm{C}(9)-\mathrm{C}(7)^{1}$ | $1.372(3)$ |
| $\mathrm{F}(5)-\mathrm{C}(5)$ | $1.340(3)$ | $\mathrm{N}(1)-\mathrm{C}(10)$ | $1.338(4)$ |
| $\mathrm{F}(6)-\mathrm{C}(6)$ | $1.338(3)$ | $\mathrm{N}(1)-\mathrm{C}(13)$ | $1.349(4)$ |
| $\mathrm{C}(1)-\mathrm{C}(2)$ | $1.373(4)$ | $\mathrm{C}(10)-\mathrm{C}(11)$ | $1.345(5)$ |
| $\mathrm{C}(1)-\mathrm{C}(6)$ | $1.376(4)$ | $\mathrm{C}(11)-\mathrm{C}(12)$ | $1.405(5)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)$ | $1.371(4)$ | $\mathrm{C}(12)-\mathrm{C}(13)$ | $1.341(4)$ |
| $\mathrm{C}(3)-\mathrm{C}(4)$ | $1.376(4)$ | $\mathrm{N}(2)-\mathrm{C}(14)$ | $1.347(4)$ |
| $\mathrm{C}(4)-\mathrm{C}(5)$ | $1.371(4)$ | $\mathrm{N}(2)-\mathrm{C}(17)$ | $1.349(4)$ |
| $\mathrm{C}(5)-\mathrm{C}(6)$ | $1.357(4)$ | $\mathrm{C}(14)-\mathrm{C}(15)$ | $1.333(5)$ |
| $\mathrm{F}(7)-\mathrm{C}(7)$ | $1.340(3)$ | $\mathrm{C}(15)-\mathrm{C}(16)$ | $1.384(5)$ |
| $\mathrm{F}(8)-\mathrm{C}(8)$ | $1.340(2)$ | $\mathrm{C}(16)-\mathrm{C}(17)$ | $1.339(4)$ |
| $\mathrm{F}(9)-\mathrm{C}(9)$ | $1.339(3)$ |  |  |

$$
{ }^{1} 1-x, 1-y,-z
$$

Table S1e Selected bond angles for $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at 150 K .

| Atom - Atom - Atom | Angle $/^{\circ}$ | Atom - Atom - Atom | Angle / |
| :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)-\mathrm{C}(2)$ | $120.6(3)$ | $\mathrm{F}(7)-\mathrm{C}(7)-\mathrm{C}(9)^{1}$ | $119.8(2)$ |
| $\mathrm{F}(1)-\mathrm{C}(1)-\mathrm{C}(6)$ | $120.2(3)$ | $\mathrm{C}(8)-\mathrm{C}(7)-\mathrm{C}(9)^{1}$ | $120.3(2)$ |
| $\mathrm{C}(2)-\mathrm{C}(1)-\mathrm{C}(6)$ | $119.3(2)$ | $\mathrm{F}(8)-\mathrm{C}(8)-\mathrm{C}(7)$ | $119.9(2)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)-\mathrm{C}(1)$ | $119.9(3)$ | $\mathrm{F}(8)-\mathrm{C}(8)-\mathrm{C}(9)$ | $120.2(2)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)-\mathrm{C}(3)$ | $119.4(3)$ | $\mathrm{C}(7)-\mathrm{C}(8)-\mathrm{C}(9)$ | $119.9(2)$ |
| $\mathrm{C}(3)-\mathrm{C}(2)-\mathrm{C}(1)$ | $120.6(2)$ | $\mathrm{F}(9)-\mathrm{C}(9)-\mathrm{C}(7)^{1}$ | $120.4(2)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)-\mathrm{C}(2)$ | $120.6(3)$ | $\mathrm{F}(9)-\mathrm{C}(9)-\mathrm{C}(8)$ | $119.7(2)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)-\mathrm{C}(4)$ | $119.9(3)$ | $\mathrm{C}(7)^{1}-\mathrm{C}(9)-\mathrm{C}(8)$ | $119.9(2)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)-\mathrm{C}(4)$ | $119.5(2)$ | $\mathrm{C}(10)-\mathrm{N}(1)-\mathrm{C}(13)$ | $109.8(2)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)-\mathrm{C}(3)$ | $120.0(2)$ | $\mathrm{N}(1)-\mathrm{C}(10)-\mathrm{C}(11)$ | $107.7(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)-\mathrm{C}(5)$ | $120.3(2)$ | $\mathrm{C}(10)-\mathrm{C}(11)-\mathrm{C}(12)$ | $107.7(3)$ |
| $\mathrm{C}(5)-\mathrm{C}(4)-\mathrm{C}(3)$ | $119.7(2)$ | $\mathrm{C}(13)-\mathrm{C}(12)-\mathrm{C}(11)$ | $106.7(3)$ |
| $\mathrm{F}(5)-\mathrm{C}(5)-\mathrm{C}(4)$ | $119.3(2)$ | $\mathrm{C}(12)-\mathrm{C}(13)-\mathrm{N}(1)$ | $108.1(3)$ |
| $\mathrm{F}(5)-\mathrm{C}(5)-\mathrm{C}(6)$ | $120.0(2)$ | $\mathrm{C}(14)-\mathrm{N}(2)-\mathrm{C}(17)$ | $108.7(3)$ |
| $\mathrm{C}(6)-\mathrm{C}(5)-\mathrm{C}(4)$ | $120.6(2)$ | $\mathrm{C}(15)-\mathrm{C}(14)-\mathrm{N}(2)$ | $108.1(3)$ |
| $\mathrm{F}(6)-\mathrm{C}(6)-\mathrm{C}(1)$ | $120.0(2)$ | $\mathrm{C}(14)-\mathrm{C}(15)-\mathrm{C}(16)$ | $107.8(3)$ |
| $\mathrm{F}(6)-\mathrm{C}(6)-\mathrm{C}(5)$ | $119.8(2)$ | $\mathrm{C}(17)-\mathrm{C}(16)-\mathrm{C}(15)$ | $107.3(3)$ |
| $\mathrm{C}(5)-\mathrm{C}(6)-\mathrm{C}(1)$ | $120.2(2)$ | $\mathrm{C}(16)-\mathrm{C}(17)-\mathrm{N}(2)$ | $108.1(3)$ |
| $\mathrm{F}(7)-\mathrm{C}(7)-\mathrm{C}(8)$ | $119.9(2)$ |  |  |

Table S2a Crystal data and structure refinement for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K .

| Identification code | exp_299 |
| :---: | :---: |
| Empirical formula | $\mathrm{C}_{11} \mathrm{H}_{5} \mathrm{~F}_{6} \mathrm{~N}$ |
| Formula weight | 265.16 |
| Temperature / K | 150 |
| Crystal system | orthorhombic |
| Space group | $P 2{ }_{1}{ }_{1} 2_{1}$ |
| $a / \AA$ | 5.85770(10) |
| $b / \AA$ | 10.4534(2) |
| $c / \AA$ | 17.3482(3) |
| $\alpha /{ }^{\circ}$ | 90 |
| $\beta /{ }^{\circ}$ | 90 |
| $\gamma /{ }^{\circ}$ | 90 |
| Volume / $\AA^{3}$ | 1062.28(3) |
| Z | 4 |
| $\rho_{\text {calc }} / \mathrm{g} \mathrm{cm}^{-3}$ | 1.658 |
| $\mu / \mathrm{mm}^{-1}$ | 1.564 |
| $F(000)$ | 528.0 |
| Crystal size / mm ${ }^{3}$ | $0.862 \times 0.218 \times 0.212$ |
| Radiation | Cu K $\alpha$ ( $\lambda=1.54184 \AA$ ) |
| $2 \theta$ range for data collection / ${ }^{\circ}$ | 9.878 to 159.14 |
| Index ranges | $-7 \leq h \leq 6,-13 \leq k \leq 13,-21 \leq l \leq 21$ |
| Reflections collected | 24728 |
| Independent reflections | 2275 [ $\left.R_{\text {int }}=0.0615, R_{\text {sigma }}=0.0240\right]$ |
| Data/restraints/parameters | 2275/0/184 |
| Goodness-of-fit on $F^{2}$ | 1.065 |
| Final $R$ indexes [ $I \geq 2 \sigma(I)$ ] | $R_{1}=0.0346, w R_{2}=0.0929$ |
| Final $R$ indexes [all data] | $R_{1}=0.0429, w R_{2}=0.0983$ |
| Largest diff. peak/hole / e $\AA^{-3}$ | 0.13/-0.15 |
| Flack parameter | 0.06(7) |

Table S2b Fractional atomic coordinates and equivalent isotropic displacement parameters for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at $150 \mathrm{~K} . U_{\text {eq }}$ is defined as $1 / 3$ of the trace of the orthogonalised $U_{\mathrm{ij}}$ tensor.

| Atom | $\boldsymbol{x}$ | $\boldsymbol{y}$ | $\boldsymbol{z}$ | $\boldsymbol{U}(\mathrm{eq}) / \AA^{2}$ |
| :--- | :--- | :--- | :--- | :--- |
| $\mathrm{~F}(1)$ | $0.5109(4)$ | $0.4753(16)$ | $0.8221(10)$ | $0.0678(6)$ |
| $\mathrm{F}(2)$ | $0.2028(3)$ | $0.4642(18)$ | $0.7056(10)$ | $0.0672(5)$ |
| $\mathrm{F}(3)$ | $0.2788(3)$ | $0.3088(19)$ | $0.5835(9)$ | $0.0680(6)$ |
| $\mathrm{F}(4)$ | $0.6658(4)$ | $0.1678(17)$ | $0.5764(9)$ | $0.0693(6)$ |
| $\mathrm{F}(5)$ | $0.9736(3)$ | $0.1779(17)$ | $0.6932(11)$ | $0.0650(5)$ |
| $\mathrm{F}(6)$ | $0.8959(3)$ | $0.3323(18)$ | $0.8155(9)$ | $0.0645(5)$ |
| $\mathrm{C}(1)$ | $0.5491(5)$ | $0.3993(2)$ | $0.7614(14)$ | $0.0446(6)$ |
| $\mathrm{C}(2)$ | $0.3932(5)$ | $0.3930(2)$ | $0.7027(14)$ | $0.0452(6)$ |
| $\mathrm{C}(3)$ | $0.4307(5)$ | $0.3146(3)$ | $0.6408(14)$ | $0.0458(6)$ |
| $\mathrm{C}(4)$ | $0.6264(5)$ | $0.2428(2)$ | $0.6371(13)$ | $0.0454(6)$ |
| $\mathrm{C}(5)$ | $0.7833(5)$ | $0.2485(2)$ | $0.6959(14)$ | $0.0446(6)$ |
| $\mathrm{C}(6)$ | $0.7445(5)$ | $0.3271(2)$ | $0.7583(13)$ | $0.0438(6)$ |
| $\mathrm{N}(1)$ | $0.5647(4)$ | $0.6745(2)$ | $0.6118(13)$ | $0.0537(6)$ |
| $\mathrm{C}(7)$ | $0.4135(5)$ | $0.6625(3)$ | $0.5545(16)$ | $0.0483(6)$ |
| $\mathrm{C}(8)$ | $0.4437(5)$ | $0.5830(3)$ | $0.4922(15)$ | $0.0481(6)$ |
| $\mathrm{C}(9)$ | $0.6395(6)$ | $0.5119(3)$ | $0.4879(16)$ | $0.0505(6)$ |
| $\mathrm{C}(10)$ | $0.7985(5)$ | $0.5224(3)$ | $0.5457(18)$ | $0.0516(7)$ |
| $\mathrm{C}(11)$ | $0.7549(6)$ | $0.6045(3)$ | $0.6058(16)$ | $0.0525(7)$ |
| $\mathrm{H}(7)$ | $0.281(6)$ | $0.713(3)$ | $0.5609(19)$ | $0.0690(10)$ |
| $\mathrm{H}(8)$ | $0.325(6)$ | $0.579(3)$ | $0.4552(17)$ | $0.0580(9)$ |
| $\mathrm{H}(9)$ | $0.670(6)$ | $0.459(3)$ | $0.4438(16)$ | $0.0610(9)$ |
| $\mathrm{H}(10)$ | $0.941(6)$ | $0.476(3)$ | $0.5472(18)$ | $0.0600(9)$ |
| $\mathrm{H}(11)$ | $0.863(6)$ | $0.612(3)$ | $0.6501(19)$ | $0.0660(9)$ |

Table S2c Anisotropic displacement parameters for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K . The anisotropic displacement factor exponent has the form: $-2 \pi^{2}\left[h^{2} a^{* 2} U_{11}+2 h k a^{*} b^{*} U_{12}+\ldots\right]$.

| Atom | $\boldsymbol{U}_{11} / \AA^{2}$ | $\boldsymbol{U}_{22} / \AA^{2}$ | $\boldsymbol{U}_{33} / \AA^{2}$ | $\boldsymbol{U}_{23} / \AA^{2}$ | $\boldsymbol{U}_{13} / \AA^{2}$ | $\boldsymbol{U}_{12} / \AA^{2}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{~F}(1)$ | $0.0968(16)$ | $0.0557(9)$ | $0.0510(8)$ | $-0.0149(7)$ | $0.0075(9)$ | $0.0084(10)$ |
| $\mathrm{F}(2)$ | $0.0495(11)$ | $0.0666(10)$ | $0.0857(12)$ | $0.0157(9)$ | $0.0070(9)$ | $0.0154(9)$ |
| $\mathrm{F}(3)$ | $0.0723(12)$ | $0.0800(11)$ | $0.0517(8)$ | $0.0157(8)$ | $-0.0230(8)$ | $-0.0248(10)$ |
| $\mathrm{F}(4)$ | $0.1043(15)$ | $0.0590(9)$ | $0.0446(8)$ | $-0.0109(7)$ | $0.0207(10)$ | $-0.0101(11)$ |
| $\mathrm{F}(5)$ | $0.0542(10)$ | $0.0621(10)$ | $0.0788(11)$ | $0.0134(9)$ | $0.0197(9)$ | $0.0156(9)$ |
| $\mathrm{F}(6)$ | $0.0647(11)$ | $0.0762(11)$ | $0.0525(8)$ | $0.0105(8)$ | $-0.0194(8)$ | $-0.0061(10)$ |
| $\mathrm{C}(1)$ | $0.0589(17)$ | $0.0371(11)$ | $0.0380(11)$ | $-0.0018(9)$ | $0.0037(11)$ | $-0.0013(11)$ |
| $\mathrm{C}(2)$ | $0.0385(14)$ | $0.0448(12)$ | $0.0524(13)$ | $0.0092(11)$ | $0.0054(12)$ | $0.0017(11)$ |
| $\mathrm{C}(3)$ | $0.0485(16)$ | $0.0500(13)$ | $0.0389(11)$ | $0.0102(10)$ | $-0.0042(10)$ | $-0.0140(12)$ |
| $\mathrm{C}(4)$ | $0.0619(17)$ | $0.0405(11)$ | $0.0337(10)$ | $-0.0006(9)$ | $0.0119(11)$ | $-0.0093(12)$ |
| $\mathrm{C}(5)$ | $0.0433(15)$ | $0.0422(11)$ | $0.0483(12)$ | $0.0096(10)$ | $0.0114(11)$ | $0.0053(11)$ |
| $\mathrm{C}(6)$ | $0.0469(15)$ | $0.0466(12)$ | $0.0378(11)$ | $0.0062(10)$ | $-0.0060(10)$ | $-0.0045(12)$ |
| $\mathrm{N}(1)$ | $0.0624(16)$ | $0.0486(12)$ | $0.0502(12)$ | $-0.0042(10)$ | $0.0035(11)$ | $0.0005(12)$ |
| $\mathrm{C}(7)$ | $0.0430(15)$ | $0.0456(13)$ | $0.0565(14)$ | $0.0061(11)$ | $0.0057(12)$ | $0.0046(12)$ |
| $\mathrm{C}(8)$ | $0.0478(16)$ | $0.0476(13)$ | $0.0487(13)$ | $0.0053(11)$ | $-0.0045(12)$ | $-0.0074(11)$ |
| $\mathrm{C}(9)$ | $0.0584(18)$ | $0.0429(13)$ | $0.0502(13)$ | $-0.0010(11)$ | $0.0114(13)$ | $-0.0041(12)$ |
| $\mathrm{C}(10)$ | $0.0404(16)$ | $0.0452(13)$ | $0.0691(16)$ | $0.0116(12)$ | $0.0078(13)$ | $0.0027(12)$ |
| $\mathrm{C}(11)$ | $0.0504(17)$ | $0.0524(14)$ | $0.0546(14)$ | $0.0091(12)$ | $-0.0085(13)$ | $-0.0109(13)$ |

Table S2d Selected bond lengths for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K .

| Atom - Atom | Length / $\AA$ | Atom - Atom | Length / $\AA$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)$ | $1.337(3)$ | $\mathrm{C}(3)-\mathrm{C}(4)$ | $1.372(4)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)$ | $1.342(3)$ | $\mathrm{C}(4)-\mathrm{C}(5)$ | $1.374(4)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)$ | $1.336(3)$ | $\mathrm{C}(5)-\mathrm{C}(6)$ | $1.377(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)$ | $1.334(3)$ | $\mathrm{N}(1)-\mathrm{C}(7)$ | $1.337(4)$ |
| $\mathrm{F}(5)-\mathrm{C}(5)$ | $1.338(3)$ | $\mathrm{N}(1)-\mathrm{C}(11)$ | $1.338(4)$ |
| $\mathrm{F}(6)-\mathrm{C}(6)$ | $1.332(3)$ | $\mathrm{C}(7)-\mathrm{C}(8)$ | $1.374(4)$ |
| $\mathrm{C}(1)-\mathrm{C}(2)$ | $1.370(4)$ | $\mathrm{C}(8)-\mathrm{C}(9)$ | $1.369(4)$ |
| $\mathrm{C}(1)-\mathrm{C}(6)$ | $1.372(4)$ | $\mathrm{C}(9)-\mathrm{C}(10)$ | $1.374(4)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)$ | $1.368(4)$ | $\mathrm{C}(10)-\mathrm{C}(11)$ | $1.374(4)$ |

Table S2e Selected bond angles for $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K .

| Atom - Atom - Atom | Angle $\boldsymbol{~}^{\circ}$ | Atom - Atom - Atom | Angle $\boldsymbol{~}^{\circ}$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)-\mathrm{C}(2)$ | $120.2(2)$ | $\mathrm{F}(5)-\mathrm{C}(5)-\mathrm{C}(4)$ | $120.5(2)$ |
| $\mathrm{F}(1)-\mathrm{C}(1)-\mathrm{C}(6)$ | $119.9(2)$ | $\mathrm{F}(5)-\mathrm{C}(5)-\mathrm{C}(6)$ | $119.6(2)$ |
| $\mathrm{C}(2)-\mathrm{C}(1)-\mathrm{C}(6)$ | $120.0(2)$ | $\mathrm{C}(4)-\mathrm{C}(5)-\mathrm{C}(6)$ | $119.9(2)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)-\mathrm{C}(1)$ | $119.9(2)$ | $\mathrm{F}(6)-\mathrm{C}(6)-\mathrm{C}(1)$ | $120.2(2)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)-\mathrm{C}(3)$ | $119.7(2)$ | $\mathrm{F}(6)-\mathrm{C}(6)-\mathrm{C}(5)$ | $120.0(2)$ |
| $\mathrm{C}(3)-\mathrm{C}(2)-\mathrm{C}(1)$ | $120.4(2)$ | $\mathrm{C}(1)-\mathrm{C}(6)-\mathrm{C}(5)$ | $119.8(2)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)-\mathrm{C}(2)$ | $120.3(3)$ | $\mathrm{C}(7)-\mathrm{N}(1)-\mathrm{C}(11)$ | $116.3(2)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)-\mathrm{C}(4)$ | $119.8(2)$ | $\mathrm{N}(1)-\mathrm{C}(7)-\mathrm{C}(8)$ | $123.8(3)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)-\mathrm{C}(4)$ | $119.9(2)$ | $\mathrm{C}(9)-\mathrm{C}(8)-\mathrm{C}(7)$ | $118.7(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)-\mathrm{C}(3)$ | $120.2(3)$ | $\mathrm{C}(8)-\mathrm{C}(9)-\mathrm{C}(10)$ | $119.0(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)-\mathrm{C}(5)$ | $119.7(3)$ | $\mathrm{C}(11)-\mathrm{C}(10)-\mathrm{C}(9)$ | $118.5(3)$ |
| $\mathrm{C}(3)-\mathrm{C}(4)-\mathrm{C}(5)$ | $120.1(2)$ | $\mathrm{N}(1)-\mathrm{C}(11)-\mathrm{C}(10)$ | $123.8(3)$ |

Table S3a Crystal data and structure refinement for phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 160 K .

| Identification code | exp_3163 |
| :---: | :---: |
| Empirical formula | $\mathrm{C}_{14} \mathrm{H}_{11} \mathrm{~F}_{5}$ |
| Formula weight | 274.23 |
| Temperature / K | 160 |
| Crystal system | triclinic |
| Space group | $P \overline{1}$ |
| $a / \AA$ | 6.3499(4) |
| $b / \AA$ | 7.2178(5) |
| $c / \AA$ | 7.3804(5) |
| $\alpha /{ }^{\circ}$ | 108.087(7) |
| $\beta /{ }^{\circ}$ | 93.949(6) |
| $\gamma /{ }^{\circ}$ | 98.796(6) |
| Volume / $\AA^{3}$ | 315.32(4) |
| Z | 1 |
| $\rho_{\text {calc }} / \mathrm{g} \mathrm{cm}^{-3}$ | 1.444 |
| $\mu / \mathrm{mm}^{-1}$ | 1.191 |
| $F(000)$ | 140.0 |
| Crystal size / mm ${ }^{3}$ | $0.95 \times 0.394 \times 0.38$ |
| Radiation | Cu K $\alpha$ ( $\lambda=1.54184 \AA$ A |
| $2 \theta$ range for data collection / ${ }^{\circ}$ | 14.224 to 144.066 |
| Index ranges | $-7 \leq h \leq 7,-8 \leq k \leq 8,-8 \leq l \leq 7$ |
| Reflections collected | 3974 |
| Independent reflections | $1201\left[R_{\text {int }}=0.0629, R_{\text {sigma }}=0.0412\right]$ |
| Data/restraints/parameters | 1201/61/121 |
| Goodness-of-fit on $F^{2}$ | 1.156 |
| Final $R$ indexes [ $I \geq 2 \sigma(I)$ ] | $R_{1}=0.0617, w R_{2}=0.1742$ |
| Final $R$ indexes [all data] | $R_{1}=0.0692, w R_{2}=0.1894$ |
| Largest diff. peak/hole / e $\AA^{-3}$ | 0.20/-0.20 |

Table S3b Fractional atomic coordinates and equivalent isotropic displacement parameters for phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 160 K . $U_{\text {eq }}$ is defined as $1 / 3$ of the trace of the orthogonalised $U_{\mathrm{ij}}$ tensor.

| Atom | $\boldsymbol{x}$ | $\boldsymbol{y}$ | $\boldsymbol{z}$ | $\boldsymbol{U}(\mathrm{eq}) / \AA^{2}$ | Occupancy |
| :--- | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)$ | $-0.4012(2)$ | $-0.1608(3)$ | $0.3161(2)$ | $0.0791(8)$ | $0.972(5)$ |
| $\mathrm{F}(2)$ | $-0.2751(4)$ | $0.2246(4)$ | $0.4284(3)$ | $0.0811(10)$ | $0.697(5)$ |
| $\mathrm{F}(3)$ | $0.1330(4)$ | $0.3918(3)$ | $0.6222(3)$ | $0.0847(9)$ | $0.830(6)$ |
| $\mathrm{C}(1)$ | $-0.2040(4)$ | $-0.0812(4)$ | $0.4065(3)$ | $0.0571(7)$ | 1 |
| $\mathrm{C}(2)$ | $-0.1368(4)$ | $0.1189(4)$ | $0.4663(3)$ | $0.0568(7)$ | 1 |
| $\mathrm{C}(3)$ | $0.0655(4)$ | $0.1997(4)$ | $0.5599(3)$ | $0.0598(7)$ | 1 |
| $\mathrm{C}(4)$ | $0.2014(3)$ | $-0.0151(4)$ | $0.0645(3)$ | $0.0543(7)$ | 1 |
| $\mathrm{C}(5)$ | $-0.0489(4)$ | $0.1811(4)$ | $0.0152(3)$ | $0.0551(7)$ | 1 |
| $\mathrm{C}(6)$ | $0.1585(3)$ | $0.1734(3)$ | $0.0824(3)$ | $0.0536(7)$ | 1 |
| $\mathrm{C}(7)$ | $0.3250(5)$ | $0.3560(5)$ | $0.1681(5)$ | $0.0832(10)$ | 1 |
| $\mathrm{H}(1)^{\dagger}+$ | $=\boldsymbol{x}_{\mathrm{F}(1)}$ | $=\boldsymbol{y}_{\mathrm{F}(1)}$ | $=\boldsymbol{z}_{\mathrm{F}(1)}$ | $=\boldsymbol{U}_{\mathrm{F}(1)}$ | $=1-O_{\mathrm{F}(1)}$ |
| $\mathrm{H}(2) \dagger$ | $=\boldsymbol{x}_{\mathrm{F}(2)}$ | $=\boldsymbol{y}_{\mathrm{F}(2)}$ | $=\boldsymbol{z}_{\mathrm{F}(2)}$ | $=\boldsymbol{U}_{\mathrm{F}(2)}$ | $=1-O_{\mathrm{F}(2)}$ |
| $\mathrm{H}(3) \dagger$ | $=\boldsymbol{x}_{\mathrm{F}(3)}$ | $=\boldsymbol{y}_{\mathrm{F}(3)}$ | $=\boldsymbol{z}_{\mathrm{F}(3)}$ | $=\boldsymbol{U}_{\mathrm{F}(3)}$ | $=1-O_{\mathrm{F}(3)}$ |
| $\mathrm{H}(4)$ | $0.343(5)$ | $-0.016(4)$ | $0.103(4)$ | $0.0720(8)$ | 1 |
| $\mathrm{H}(5)$ | $-0.076(5)$ | $0.313(5)$ | $0.027(5)$ | $0.0810(9)$ | 1 |
| $\mathrm{H}(7 \mathrm{~A})$ | $0.444(10)$ | $0.365(9)$ | $0.091(9)$ | 0.125 | $1 / 2$ |
| $\mathrm{H}(7 \mathrm{~B})$ | $0.394(13)$ | $0.369(9)$ | $0.296(8)$ | 0.125 | $1 / 2$ |
| $\mathrm{H}(7 \mathrm{C})$ | $0.271(7)$ | $0.480(4)$ | $0.186(15)$ | 0.125 | $1 / 2$ |
| $\mathrm{H}(7 \mathrm{D})$ | $0.295(10)$ | $0.444(9)$ | $0.291(9)$ | 0.125 | $1 / 2$ |
| $\mathrm{H}(7 \mathrm{E})$ | $0.345(13)$ | $0.440(9)$ | $0.086(8)$ | 0.125 | $1 / 2$ |
| $\mathrm{H}(7 \mathrm{~F})$ | $0.469(5)$ | $0.329(5)$ | $0.196(15)$ | 0.125 | $1 / 2$ |

† The atomic coordinates and atomic displacement parameters of $H(1), H(2)$, and $H(3)$ are tied to those of $F(1), F(2)$, and $F(3)$, respectively. The site occupancies of $F(1)$ plus $H(1), F(2)$ plus $H(2)$, and $F(3)$ plus $H(3)$ are tied to $100 \%$ occupancy.

Table S3c Anisotropic displacement parameters for phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 160 K . The anisotropic displacement factor exponent has the form: $-2 \pi^{2}\left[h^{2} a^{* 2} U_{11}+2 h k a^{*} b^{*} U_{12}+\ldots\right]$.

| Atom | $\boldsymbol{U}_{11} / \AA^{2}$ | $\boldsymbol{U}_{22} / \AA^{2}$ | $\boldsymbol{U}_{33} / \AA^{2}$ | $\boldsymbol{U}_{23} / \AA^{2}$ | $\boldsymbol{U}_{13} / \AA^{2}$ | $\boldsymbol{U}_{12} / \AA^{2}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{~F}(1)$ | $0.0541(10)$ | $0.1125(15)$ | $0.0688(11)$ | $0.0262(9)$ | $0.0040(7)$ | $0.0182(8)$ |
| $\mathrm{F}(2)$ | $0.0891(17)$ | $0.1013(19)$ | $0.0777(16)$ | $0.0398(13)$ | $0.0246(11)$ | $0.0626(14)$ |
| $\mathrm{F}(3)$ | $0.0999(16)$ | $0.0670(13)$ | $0.0919(16)$ | $0.0249(10)$ | $0.0192(11)$ | $0.0282(10)$ |
| $\mathrm{C}(1)$ | $0.0494(12)$ | $0.0864(17)$ | $0.0434(11)$ | $0.0243(11)$ | $0.0132(9)$ | $0.0264(11)$ |
| $\mathrm{C}(2)$ | $0.0574(13)$ | $0.0824(16)$ | $0.0481(12)$ | $0.0306(11)$ | $0.0187(9)$ | $0.0401(11)$ |
| $\mathrm{C}(3)$ | $0.0675(15)$ | $0.0680(16)$ | $0.0497(13)$ | $0.0189(11)$ | $0.0193(10)$ | $0.0256(12)$ |
| $\mathrm{C}(4)$ | $0.0450(11)$ | $0.0859(17)$ | $0.0453(12)$ | $0.0280(11)$ | $0.0117(8)$ | $0.0346(11)$ |
| $\mathrm{C}(5)$ | $0.0616(14)$ | $0.0681(14)$ | $0.0521(13)$ | $0.0282(10)$ | $0.0210(10)$ | $0.0373(12)$ |
| $\mathrm{C}(6)$ | $0.0545(13)$ | $0.0696(14)$ | $0.0419(11)$ | $0.0182(10)$ | $0.0162(9)$ | $0.0226(10)$ |
| $\mathrm{C}(7)$ | $0.0791(19)$ | $0.0890(2)$ | $0.0702(18)$ | $0.0127(15)$ | $0.0180(14)$ | $0.0063(16)$ |

Table S3d $\quad$ Selected bond lengths for phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 160 K .

| Atom - Atom | Length / A | Atom - Atom | Length / Å |
| :---: | :---: | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)$ | $1.330(3)$ | $\mathrm{C}(3)-\mathrm{C}(1)^{1}$ | $1.378(4)$ |
| $\mathrm{F}(2)-\mathrm{C}(2)$ | $1.318(3)$ | $\mathrm{C}(4)-\mathrm{C}(5)^{2}$ | $1.360(4)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)$ | $1.309(3)$ | $\mathrm{C}(4)-\mathrm{C}(6)$ | $1.397(3)$ |
| $\mathrm{C}(1)-\mathrm{C}(2)$ | $1.361(4)$ | $\mathrm{C}(5)-\mathrm{C}(4)^{2}$ | $1.360(4)$ |
| $\mathrm{C}(1)-\mathrm{C}(3)^{1}$ | $1.378(4)$ | $\mathrm{C}(5)-\mathrm{C}(6)$ | $1.390(3)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)$ | $1.366(4)$ | $\mathrm{C}(6)-\mathrm{C}(7)$ | $1.490(4)$ |

${ }^{1}-x,-y, 1-z ;{ }^{2}-x,-y,-z$

Table S3e $\quad$ Selected bond angles for phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 160 K .

| Atom - Atom - Atom | Angle / ${ }^{\circ}$ | Atom - Atom - Atom | Angle / ${ }^{\circ}$ |
| :---: | :---: | :---: | :---: |
| $F(1)-C(1)-C(2)$ | 120.1(2) | $F(3)-C(3)-C(2)$ | 121.1(2) |
| $F(1)-C(1)-C(3)^{1}$ | 120.6(3) | $C(2)-C(3)-C(1)^{1}$ | 120.9(3) |
| $C(2)-C(1)-C(3)^{1}$ | 119.2(2) | $C(5)^{2}-C(4)-C(6)$ | 121.6(2) |
| $F(2)-C(2)-C(1)$ | 116.6(3) | $C(4)^{2}-C(5)-C(6)$ | 122.2(2) |
| $F(2)-C(2)-C(3)$ | 123.5(3) | $C(4)-C(6)-C(7)$ | 122.1(2) |
| $C(1)-C(2)-C(3)$ | 119.9(2) | $C(5)-C(6)-C(4)$ | 116.1(2) |
| $F(3)-C(3)-C(1)^{1}$ | 117.9(3) | $C(5)-C(6)-C(7)$ | 121.8(2) |
| ${ }^{1}-x,-y, 1-z ;{ }^{2}-x,-y,-z$ |  |  |  |

Table S4a Crystal data and structure refinement for phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 120 K .

| Identification code | exp_3166 |
| :---: | :---: |
| Empirical formula | $\mathrm{C}_{14} \mathrm{H}_{11} \mathrm{~F}_{5}$ |
| Formula weight | 274.23 |
| Temperature / K | 120.5(2) |
| Crystal system | triclinic |
| Space group | $\overline{1}$ |
| $a / \AA$ | 10.2296(5) |
| $b / \AA$ | 8.7819(5) |
| $c / \AA$ | 14.6776(7) |
| $\alpha /{ }^{\circ}$ | 106.811(5) |
| $\beta /{ }^{\circ}$ | 80.111(4) |
| $\gamma /{ }^{\circ}$ | 97.168(4) |
| Volume / $\AA^{3}$ | 1239.51(12) |
| Z | 4 |
| $\rho_{\text {calc }} / \mathrm{g} \mathrm{cm}^{-3}$ | 1.469 |
| $\mu / \mathrm{mm}^{-1}$ | 1.212 |
| $F(000)$ | 560.0 |
| Crystal size / mm ${ }^{3}$ | $0.955 \times 0.383 \times 0.365$ |
| Radiation | $\mathrm{CuK} \alpha(\lambda=1.54184 \AA$ ) |
| $2 \theta$ range for data collection $/^{\circ}$ | 10.1 to 145.608 |
| Index ranges | $-12 \leq h \leq 12,-10 \leq k \leq 10,-17 \leq l \leq 15$ |
| Reflections collected | 9202 |
| Independent reflections | $2405\left[R_{\text {int }}=0.0516, R_{\text {sigma }}=0.0312\right]$ |
| Data/restraints/parameters | 2405/8/204 |
| Goodness-of-fit on $F^{2}$ | 1.092 |
| Final $R$ indexes [ $I \geq 2 \sigma(I)$ ] | $R_{1}=0.0685, w R_{2}=0.2010$ |
| Final $R$ indexes [all data] | $R_{1}=0.0770, w R_{2}=0.2184$ |
| Largest diff. peak/hole / e $\AA^{-3}$ | 0.78/-0.76 |

Table S4b Fractional atomic coordinates and equivalent isotropic displacement parameters for phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 120 K . $U_{\text {eq }}$ is defined as $1 / 3$ of the trace of the orthogonalised $U_{\mathrm{ij}}$ tensor.

| Atom | $x$ | $y$ | $z$ | $U(\mathrm{eq}) /$ A $^{2}$ | Occupancy |
| :---: | :---: | :---: | :---: | :---: | :---: |
| F(1) | 0.1222(16) | 0.2716(19) | 0.3379(11) | 0.0556(5) | 1 |
| F(2) | 0.2523(6) | 0.0057(3) | 0.2786(3) | 0.0770(4) | 0.224(6) |
| F(3) | 0.1369(15) | -0.2759(2) | 0.1828(12) | 0.0527(5) | 0.925(2) |
| F(4) | -0.1166(14) | -0.2922(17) | 0.1520(10) | 0.0479(4) | 1 |
| F(5) | -0.2524(13) | -0.0278(2) | 0.2140(11) | 0.0493(5) | $=O_{\mathrm{F}(3)}$ |
| F(6) | -0.1312(16) | 0.2535(2) | 0.3092(12) | 0.0543(5) | $=O_{\mathrm{F}(3)}$ |
| C(1) | 0.0641(2) | 0.1312(3) | 0.2927(15) | 0.0383(6) | 1 |
| C(2) | 0.1338(2) | -0.0021(3) | 0.2619(14) | 0.0396(6) | 1 |
| C(3) | 0.0728(2) | -0.1429(3) | 0.2151(16) | 0.0379(6) | 1 |
| C(4) | -0.0578(19) | -0.1525(3) | 0.1985(14) | 0.0338(5) | 1 |
| C(5) | -0.1266(19) | -0.0186(3) | 0.2302(15) | 0.0356(6) | 1 |
| C(6) | -0.0666(2) | 0.1242(3) | 0.2772(16) | 0.0387(6) | 1 |
| C(7) | 0.1179(2) | -0.0624(3) | 0.4974(16) | 0.0362(5) | 1 |
| C(8) | 0.1051(2) | 0.0999(3) | 0.5364(15) | 0.0358(5) | 1 |
| C(9) | -0.0133(2) | 0.1653(3) | 0.5398(15) | 0.0348(5) | 1 |
| $\mathrm{C}(10)$ | -0.0278(3) | 0.3408(3) | 0.5817(2) | 0.0501(7) | 1 |
| C(11) | $0.1156(2)$ | 0.0835(3) | 0.0282(15) | 0.0372(6) | 1 |
| C(12) | -0.1126(2) | 0.0785(3) | 0.0144(16) | 0.0374(6) | 1 |
| C(13) | 0.0027(2) | 0.1659(3) | 0.0441(15) | 0.0364(5) | 1 |
| C(14) | 0.0061(4) | 0.3424(3) | 0.0901(2) | 0.0655(9) | 1 |
| H(2) $\ddagger$ | 0.2179(3) | 0.0035(3) | 0.2740(2) | 0.047 | $=1-O_{\mathrm{F}(2)}$ |
| H(3) ${ }^{+}$ | $=\boldsymbol{x}_{\mathrm{F}(3)}$ | $=y_{\text {F(3) }}$ | $=\boldsymbol{z}_{F(3)}$ | 0.045 | $=1-O_{F(3)}$ |
| H(5) ${ }^{+}$ | $=x_{\text {F }}(5)$ | $=y_{\text {F }}(5)$ | $=\boldsymbol{z}_{\mathrm{F}(5)}$ | 0.043 | $=1-O_{\text {F(5) }}$ |
| $\mathrm{H}(6){ }^{+}$ | $=x_{\text {F }(6)}$ | $=y_{\text {F(6) }}$ | $=\boldsymbol{z}_{\text {F(6) }}$ | 0.046 | $=1-O_{\text {F(6) }}$ |
| H(7) | 0.200(3) | -0.105(3) | 0.500(2) | 0.0500(8) | 1 |
| H(8) | 0.177(3) | 0.173(3) | 0.5626(19) | 0.0400(7) | 1 |
| H(10A) | -0.1046 | 0.3697 | 0.5604 | 0.0750 | 1 |
| H(10B) | 0.0530 | 0.4010 | 0.5602 | 0.0750 | 1 |
| H(10C) | -0.0413 | 0.3662 | 0.6521 | 0.0750 | 1 |
| H(11) | 0.196(3) | 0.142(3) | 0.050(2) | 0.0520(8) | 1 |
| H(12) | -0.186(3) | 0.141(4) | 0.023(2) | 0.0660(9) | 1 |
| H(14A) | -0.0133 | 0.3628 | 0.1601 | 0.0980 | 1 |
| H(14B) | -0.0609 | 0.3878 | 0.0675 | 0.0980 | 1 |
| H(14C) | 0.0946 | 0.3920 | 0.0728 | 0.0980 | 1 |

+ The atomic coordinates $H(3), H(5)$, and $H(6)$ are tied to those of $F(3), F(5)$, and $F(6)$, respectively. The site occupancies of $\mathrm{F}(3)$ plus $\mathrm{H}(3), \mathrm{F}(5)$ plus $\mathrm{H}(5)$, and $\mathrm{F}(6)$ plus $\mathrm{H}(6)$ are tied to $100 \%$ occupancy. $\ddagger$ The atomic coordinates of $\mathrm{H}(2)$ are restrained such that $\mathrm{H}(2)$ lies on the $\mathrm{C}(2)-\mathrm{F}(2)$ bond.

Table S4c Anisotropic displacement parameters for phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 120 K . The anisotropic displacement factor exponent has the form: $-2 \pi^{2}\left[h^{2} a^{* 2} U_{11}+2 h k a^{*} b^{*} U_{12}+\ldots\right]$.

| Atom | $\boldsymbol{U}_{11} / \AA^{2}$ | $\boldsymbol{U}_{22} / \AA^{2}$ | $\boldsymbol{U}_{33} / \AA^{2}$ | $\boldsymbol{U}_{23} / \AA^{2}$ | $\boldsymbol{U}_{13} / \AA^{2}$ | $\boldsymbol{U}_{12} / \AA^{2}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{~F}(1)$ | $0.0595(9)$ | $0.0536(9)$ | $0.0465(9)$ | $0.0061(7)$ | $-0.0137(7)$ | $-0.0130(7)$ |
| $\mathrm{F}(2)$ | $0.0230(3)$ | $0.1520(9)$ | $0.0670(5)$ | $0.0530(5)$ | $-0.0110(3)$ | $-0.0110(4)$ |
| $\mathrm{F}(3)$ | $0.0452(8)$ | $0.0551(10)$ | $0.0600(11)$ | $0.0131(8)$ | $-0.0072(7)$ | $0.0190(7)$ |
| $\mathrm{F}(4)$ | $0.0474(8)$ | $0.0471(8)$ | $0.0468(9)$ | $0.0104(6)$ | $-0.0126(6)$ | $-0.0077(6)$ |
| $\mathrm{F}(5)$ | $0.0243(7)$ | $0.0746(11)$ | $0.0545(10)$ | $0.0252(8)$ | $-0.0049(6)$ | $0.0067(6)$ |
| $\mathrm{F}(6)$ | $0.0509(9)$ | $0.0538(10)$ | $0.0572(11)$ | $0.0120(8)$ | $0.0021(7)$ | $0.0226(7)$ |
| $\mathrm{C}(1)$ | $0.0405(12)$ | $0.0473(13)$ | $0.0260(11)$ | $0.0118(9)$ | $-0.0049(9)$ | $-0.0052(9)$ |
| $\mathrm{C}(2)$ | $0.0280(10)$ | $0.0605(15)$ | $0.0337(12)$ | $0.0178(10)$ | $-0.0078(8)$ | $-0.0002(9)$ |
| $\mathrm{C}(3)$ | $0.0340(11)$ | $0.0473(12)$ | $0.0342(12)$ | $0.0140(10)$ | $-0.0010(9)$ | $0.0086(9)$ |
| $\mathrm{C}(4)$ | $0.0331(11)$ | $0.0433(12)$ | $0.0256(10)$ | $0.0122(9)$ | $-0.0051(8)$ | $-0.0025(9)$ |
| $\mathrm{C}(5)$ | $0.0237(9)$ | $0.0568(14)$ | $0.0296(11)$ | $0.0188(10)$ | $-0.0017(8)$ | $0.0024(9)$ |
| $\mathrm{C}(6)$ | $0.0387(12)$ | $0.0439(12)$ | $0.0329(12)$ | $0.0123(10)$ | $0.0037(9)$ | $0.0084(9)$ |
| $\mathrm{C}(7)$ | $0.0267(9)$ | $0.0522(13)$ | $0.0316(11)$ | $0.0145(10)$ | $-0.0031(8)$ | $0.0049(8)$ |
| $\mathrm{C}(8)$ | $0.0257(10)$ | $0.0515(13)$ | $0.0312(11)$ | $0.0150(9)$ | $-0.0056(8)$ | $-0.0044(8)$ |
| $\mathrm{C}(9)$ | $0.0315(10)$ | $0.0459(12)$ | $0.0275(11)$ | $0.0136(9)$ | $0.0000(8)$ | $0.0017(9)$ |
| $\mathrm{C}(10)$ | $0.0526(14)$ | $0.0468(14)$ | $0.0476(15)$ | $0.0124(11)$ | $0.0023(11)$ | $0.0061(11)$ |
| $\mathrm{C}(11)$ | $0.0273(10)$ | $0.0526(14)$ | $0.0321(12)$ | $0.0147(10)$ | $-0.0068(8)$ | $-0.0061(9)$ |
| $\mathrm{C}(12)$ | $0.0280(10)$ | $0.0536(14)$ | $0.0339(12)$ | $0.0169(10)$ | $0.0005(8)$ | $0.0099(9)$ |
| $\mathrm{C}(13)$ | $0.0410(12)$ | $0.0394(11)$ | $0.0262(11)$ | $0.0099(9)$ | $0.0031(9)$ | $0.0031(9)$ |
| $\mathrm{C}(14)$ | $0.0950(2)$ | $0.0398(14)$ | $0.0482(16)$ | $0.0052(12)$ | $0.0125(15)$ | $0.0031(13)$ |

Table S4d Selected bond lengths for phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 120 K .

| Atom - Atom | Length / A | Atom - Atom | Length / A |
| :---: | :--- | :---: | :---: |
| $\mathrm{F}(1)-\mathrm{C}(1)$ | $1.339(3)$ | $\mathrm{C}(5)-\mathrm{C}(6)$ | $1.370(3)$ |
| $+\mathrm{H}(2)-\mathrm{C}(2)$ | 0.90 | $\mathrm{C}(7)-\mathrm{C}(8)$ | $1.387(3)$ |
| $\mathrm{F}(3)-\mathrm{C}(3)$ | $1.335(3)$ | $\mathrm{C}(7)-\mathrm{C}(9)^{1}$ | $1.391(3)$ |
| $\mathrm{F}(4)-\mathrm{C}(4)$ | $1.343(2)$ | $\mathrm{C}(8)-\mathrm{C}(9)$ | $1.392(3)$ |
| $\mathrm{F}(5)-\mathrm{C}(5)$ | $1.337(2)$ | $\mathrm{C}(9)-\mathrm{C}(7)^{1}$ | $1.391(3)$ |
| $\mathrm{F}(6)-\mathrm{C}(6)$ | $1.311(2)$ | $\mathrm{C}(9)-\mathrm{C}(10)$ | $1.501(3)$ |
| $\mathrm{C}(1)-\mathrm{C}(2)$ | $1.369(3)$ | $\mathrm{C}(11)-\mathrm{C}(12)^{2}$ | $1.378(3)$ |
| $\mathrm{C}(1)-\mathrm{C}(6)$ | $1.385(3)$ | $\mathrm{C}(11)-\mathrm{C}(13)$ | $1.390(3)$ |
| $\mathrm{C}(2)-\mathrm{C}(3)$ | $1.362(3)$ | $\mathrm{C}(12)-\mathrm{C}(11)^{2}$ | $1.378(3)$ |
| $\mathrm{C}(3)-\mathrm{C}(4)$ | $1.387(3)$ | $\mathrm{C}(12)-\mathrm{C}(13)$ | $1.388(3)$ |
| $\mathrm{C}(4)-\mathrm{C}(5)$ | $1.368(3)$ | $\mathrm{C}(13)-\mathrm{C}(14)$ | $1.500(3)$ |

${ }^{1}-x,-y, 1-z ;{ }^{2}-x,-y,-z$
$\dagger \mathrm{H}(2)$ and $\mathrm{F}(2)$ have different positions in the refinement (in contrast to $\mathrm{H}(3)$ and $\mathrm{F}(3)$, etc.). The $H(2)-C(2)$ bond length was fixed at 0.9 Å and the $F(2)-C(2)$ bond length refined to $1.267(7) A ̊$.

Table S4e $\quad$ Selected bond angles for phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ at 120 K .

| Atom | Atom - Atom | Angle / ${ }^{\circ}$ | Atom - Atom - Atom | Angle ${ }^{\circ}$ |
| :---: | :---: | :---: | :---: | :---: |
| F(1) - | $C(1)-C(2)$ | 120.1(2) | $C(4)-C(5)-C(6)$ | 120.4(2) |
| F(1) - | $C(1)-C(6)$ | 118.8(2) | $F(6)-C(6)-C(1)$ | 119.4(2) |
| $\mathrm{C}(2)$ | $C(1)-C(6)$ | 121.0(2) | $F(6)-C(6)-C(5)$ | 121.3(2) |
| H(2)/F(2) - | $C(2)-C(1)$ | 120.2(2) | $C(5)-C(6)-C(1)$ | 119.26(18) |
| H(2)/F(2) - | $C(2)-C(3)$ | 120.9(2) | $C(8)-C(7)-C(9)^{1}$ | 121.3(2) |
| C(3) - | $C(2)-C(1)$ | 118.9(2) | $C(7)-C(8)-C(9)$ | 121.0(2) |
| F(3) - | $C(3)-C(2)$ | 121.2(2) | $C(7)^{1}-C(9)-C(8)$ | 117.7(2) |
| F(3) - | $C(3)-C(4)$ | 117.77(19) | $C(7)^{1}-C(9)-C(10)$ | 121.0(2) |
| C(2) - | $C(3)-C(4)$ | 121.0(2) | $C(8)-C(9)-C(10)$ | 121.3(2) |
| F(4) - | $C(4)-C(3)$ | 120.3(2) | $C(12)^{2}-C(11)-C(13)$ | 121.2(2) |
| F(4) - | $C(4)-C(5)$ | 120.25(19) | $C(11)^{2}-C(12)-C(13)$ | 121.5(2) |
| C(5) - | $C(4)-C(3)$ | 119.41(19) | $C(11)-C(13)-C(14)$ | 121.2(2) |
| F(5) - | $C(5)-C(4)$ | 119.4(2) | $C(12)-C(13)-C(11)$ | 117.3(2) |
| $F(5)-$ | $C(5)-C(6)$ | 120.2(2) | $C(12)-C(13)-C(14)$ | 121.5(2) |
| ${ }^{1}-x,-y, 1-z ;{ }^{2}-x,-y,-z$ |  |  |  |  |

Table S5 Lattice parameters and unit cell volume of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ obtained from Le Bail fits to the VT-PXRD data shown in Fig. S14. Note that the fits for the unit cell of phase II used a unit cell with $\mathrm{Z}=4$ (as the cell is doubled along $c$ and is $I$ centred) where phase $I$ is primitive triclinic with $\mathrm{Z}=1$.

| $\mathbf{T} / \mathbf{K}$ | $\boldsymbol{a} / \AA$ | $\boldsymbol{b} / \AA$ | $\boldsymbol{c} / \AA$ | $\boldsymbol{\alpha} /{ }^{\circ}$ | $\boldsymbol{\beta} /{ }^{\circ}$ | $\boldsymbol{\gamma} /{ }^{\circ}$ | $\boldsymbol{V} / \AA^{\mathbf{3}}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 100 | $10.2147(2)$ | $8.7429(3)$ | $14.6793(6)$ | $106.459(4)$ | $80.476(5)$ | $97.045(2)$ | $1235.9(12)$ |
| 110 | $10.2278(2)$ | $8.7573(3)$ | $14.6878(7)$ | $106.585(3)$ | $80.433(5)$ | $97.059(2)$ | $1239.3(12)$ |
| 120 | $10.2345(3)$ | $8.7815(3)$ | $14.7228(13)$ | $106.825(5)$ | $80.312(6)$ | $97.146(3)$ | $1244.5(14)$ |
| 130 | $6.3086(2)$ | $7.1944(3)$ | $7.3730(7)$ | $108.410(6)$ | $93.673(8)$ | $98.548(3)$ | $311.8(6)$ |
| 140 | $6.3199(1)$ | $7.2037(2)$ | $7.3767(6)$ | $108.348(5)$ | $93.781(6)$ | $98.602(3)$ | $312.9(6)$ |
| 150 | $6.3324(1)$ | $7.2130(2)$ | $7.3691(10)$ | $108.281(5)$ | $93.831(7)$ | $98.709(3)$ | $313.6(6)$ |
| 160 | $6.3466(1)$ | $7.2210(2)$ | $7.3809(7)$ | $108.166(6)$ | $93.979(5)$ | $98.812(3)$ | $315.1(5)$ |
| 170 | $6.3614(1)$ | $7.2279(3)$ | $7.3839(10)$ | $108.006(6)$ | $94.052(8)$ | $98.944(3)$ | $316.4(7)$ |
| 180 | $6.3769(1)$ | $7.2341(2)$ | $7.3797(10)$ | $107.805(5)$ | $94.094(7)$ | $99.106(3)$ | $317.4(6)$ |
| 190 | $6.3939(1)$ | $7.2403(2)$ | $7.3765(11)$ | $107.608(5)$ | $94.179(8)$ | $99.266(3)$ | $318.6(7)$ |
| 200 | $6.4133(1)$ | $7.2465(2)$ | $7.3898(12)$ | $107.386(5)$ | $94.360(9)$ | $99.415(3)$ | $320.5(7)$ |
| 210 | $6.4336(1)$ | $7.2517(3)$ | $7.3929(5)$ | $107.137(7)$ | $94.500(7)$ | $99.586(3)$ | $322.0(7)$ |
| 220 | $6.4547(1)$ | $7.2570(3)$ | $7.3949(8)$ | $106.902(6)$ | $94.654(8)$ | $99.757(3)$ | $323.5(7)$ |
| 230 | $6.4784(1)$ | $7.2610(2)$ | $7.3915(8)$ | $106.628(6)$ | $94.791(8)$ | $99.941(3)$ | $324.9(7)$ |
| 240 | $6.5033(1)$ | $7.2621(2)$ | $7.3815(4)$ | $106.271(5)$ | $94.876(4)$ | $100.185(2)$ | $326.0(5)$ |



Figure S1 Photograph showing the experimental setup for the X-ray diffraction measurements.
Microfocus X -ray sources ( Cu and Mo ) are seen on the left-hand side, the sample capillary (which is cooled with an Oxford Instruments Cryojet5) is seen mounted in a brass stub on the goniometer head, and the Rigaku HyPix-Arc $100^{\circ}$ curved Hybrid Photon Counting (HPC) X-ray detector with "go-faster" blue lights is seen on the right-hand side of the image.


Figure S2 (a) An exemplar frame of raw X-ray data from one crystallisation attempt on $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ demonstrating that the sample in the beam is not suitable for crystal structure solution. There is no sign of diffraction spots at small $d$-spacings (rings of constant $d$ with values shown in black). Secondly, the data at low angle shows poorly resolved spots lying on rings, behaviour that is typical of many small crystallites in the beam. (b) By contrast, another exemplar frame of raw X-ray data. The raw data seen here shows clean spots out to small $d$-spacing value even though some spots from different largish crystals are accidentally overlapped.


Figure S3 A subset of peak search data with a possible unit cell with the associated lattice for it highlighted in red and blue shown using the Ewald Explorer option within the CrysAlis PRO software from Rigaku. The genuine periodicity in reciprocal space is demonstrated from the sharp peaks seen in the distribution histograms along $\mathbf{a}^{*}, \mathbf{b}^{*}$, and $\mathbf{c}^{*}$ directions on the right-hand side, which are shown in an enlarged format in the lower figure.


Figure S4 This shows the same data as in Fig. S3 but with the unindexed spots hidden to the viewer. The lattice formed by the diffraction spots is now more evident.


Figure S5 Peak search result from a full-sphere of data with unindexed spots hidden to the viewer.
The distribution histograms along $\mathbf{a}^{*}, \mathbf{b}^{*}$, and $\mathbf{c}^{*}$ directions on the right-hand side now show a clean and symmetric distribution about the mid-point.


Figure S6 Asymmetric unit from the refined crystal structure of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ in space group $P 2_{1} / n$ at 150 K showing the crystallographic labelling of the atoms. H atoms are numbered according to the label of the atom to which they are bonded. The labels of a few H atoms are omitted for clarity.


Figure S7 Using the Olex2 program suite ${ }^{3}$, a view of the asymmetric unit of the crystal structure of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right):\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ in space group $P 2_{1} / n$ at 150 K showing the refined positions and thermal displacements of the $\mathrm{C}, \mathrm{N}$, and F atoms. Positions of highest residual electron density are shown in brown and correspond to the 10 expected positions of the hydrogen atoms.
(a)

(b)


Figure S8 (a) View of the head to tail zig-zag arrangement formed by the molecules in the crystal structure of solid pyrrole as determined by Goddard et al. (1997) ${ }^{8}$ with close contacts between rings shown as dashed cyan lines; (b) the contrasting tetramer unit formed by the pyrrole molecules in the asymmetric unit of the crystal structure of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right):\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$.

[^3]

Figure S9 (a) Calculated electron density map for two molecules in $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right):\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ with areas of higher electron density presented in red and lower electron density in blue, and green is the neutral point; (b) the Hirshfeld surface for the same two molecules. For the Hirshfeld surface, closest contacts to atoms from other molecules are seen in red and voids between are shown in blue; white is the neutral point.


Figure S10 DSC data (shown as "endo up") on $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ prepared as a 3:4 molar ratio sample. The sample exhibits a sharp freezing point at 240.2 K on cooling (blue line) and no other solid-solid phase transitions down to 95 K . On heating (red line), the sample showed a broad melting transition at 266.4 K . The black vertical arrows show a transition due to the presence of a slight excess of pyrrole as a result of the high volatility of $\mathrm{C}_{6} \mathrm{~F}_{6}$.


Figure S 11 (a) View of the 0.4 mm capillary sample of $\left(\mathrm{C}_{6} \mathrm{~F}_{6}\right)_{3}:\left(\mathrm{C}_{4} \mathrm{H}_{5} \mathrm{~N}\right)_{4}$ at 150 K as used in the SXD measurement, and (b) an equivalent view of the 0.2 mm capillary sample of $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ at 150 K as used in the later experiment for comparison. Absorption corrections are based on the dimensions of the capillary and not the size of some crystal whose size cannot be determined optically. Note that X-ray capillaries are supplied with a nominal size that can vary significantly (e.g. $\pm 0.05 \mathrm{~mm}$ ) from each other within a batch.


Figure S12 Refined crystal structure of $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$ in space group $\mathrm{P} 2_{1} 2_{1} 2_{1}$ at 150 K viewed down $a$ showing the crystallographic labelling of the atoms. H atoms are numbered according to the label of the atom to which they are bonded.


Figure $\mathbf{S 1 3}$ (a) Calculated electron density map for two molecules in $\mathrm{C}_{6} \mathrm{~F}_{6}: \mathrm{C}_{5} \mathrm{H}_{5} \mathrm{~N}$; (b) the Hirshfeld surface for the same two molecules. The colour schemes are the same as those described in the caption to Fig. S9.


Figure S14 (a) VT-PXRD data on $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ with the data in 10 K steps, phase II is shown in blue, phase I is shown in red, and the liquid state is is shown in green. (b) Same data shown as a colour surface, where the range of intensity is shown as a spectral scale, from low counts in dark blue to high counts in red. The presence of a single solid-state phase transition is clearly seen in the data between 120 K and 130 K . As with $\mathrm{C}_{6} \mathrm{H}_{6-n}\left(\mathrm{CH}_{3}\right)_{n}: \mathrm{C}_{6} \mathrm{~F}_{6}$ with $\mathrm{n}=0,1,2$, or 3 , the liquid state exhibits two characteristic humps, which are visible in the higher temperature patterns. In addition, two weak peaks at about $16.0^{\circ}$ and $18.7^{\circ}$ (clearly visible in the surface plot and highlighted by the two red arrows) are attributable to a trace excess of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$.


Figure S15 Change in molecular volume relative to 100 K of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ obtained by Le Bail fits to the VT-PXRD data shown in Fig. S14; the data is presented in Table S5. There is no noticeable change in the volume of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ on passing through the phase transition, the only slight change is a very small expansion along the column axis $c$. The thermal expansion is mainly due to the increase in the distance between the columns of molecules with increasing temperature as evidenced by the change in the $a$ and $b$ values given in Table S5.


Figure S16The curves show DSC data obtained on cooling (in blue) and heating (in red) for a sample of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$. A single solid-state phase transition is observed on cooling ( 126 K ) and heating ( 129 K ). The DSC curves for this phase transition are shown on an expanded ( $\times 10$ ) scale given the intensity of the peaks due to the freezing and melting transitions at 249 K and 257 K , respectively.


Figure S17 Asymmetric unit from the refined crystal structure of phase I of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ in space group $P \overline{1}$ at 160 K showing the crystallographic labelling of the atoms. H atoms are numbered according to the label of the atom to which they are bonded; disorder of the methyl groups is represented by the use of multiple H atoms which are not labelled. Unlabelled atoms in both molecules are symmetry related by inversion symmetry (shown as open blue circles) to the labelled atoms. The dotted blue line shows the column axis formed by the alternate stacking of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ and $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ molecules.


Figure S18 Asymmetric unit from the refined crystal structure of phase II of $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}: \mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ in space group $P \overline{1}$ at 120 K showing the crystallographic labelling of the atoms. H atoms are numbered according to the label of the atom to which they are bonded. Unlabelled atoms in the $p-\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ molecules are symmetry related by inversion symmetry (shown as open blue circles) to the labelled atoms. In contrast to phase I, there is no crystallographic inversion symmetry at the centre of the $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ molecule. The dotted blue line shows the column axis formed by the alternate stacking of $p$ $\mathrm{C}_{6} \mathrm{H}_{4} \mathrm{Me}_{2}$ and $\mathrm{C}_{6} \mathrm{~F}_{5} \mathrm{H}$ molecules.


[^0]:    ${ }^{1}$ Sheldrick, G. M. (2015). Acta Cryst. A71, 3-8.
    ${ }^{2}$ Sheldrick, G. M. (2015). Acta Cryst. C71, 3-8.
    ${ }^{3}$ Dolomanov, O.V. et al. (2009), J. Appl. Cryst. 42, 339-341.
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[^1]:    ${ }^{5}$ A. Le Bail, Powder Diffr., 2005, 20, 316-326.
    ${ }^{6}$ http://www.ccp14.ac.uk/tutorial/lhpm-rietica/index.html; http://www.rietica.org/ .

[^2]:    ${ }^{7}$ Spackman, P. R. et al. (2021). J. Appl. Cryst. 54, 1006-1011.

[^3]:    ${ }^{8}$ Goddard, R., Heinemann, O. \& Krüger, C. (1997). Acta Cryst. C53, 1846-1850.

