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Supporting information for article:

Self-assembly of block copolymers under non-isothermal annealing conditions as revealed by grazing-incidence small-angle X-ray scattering

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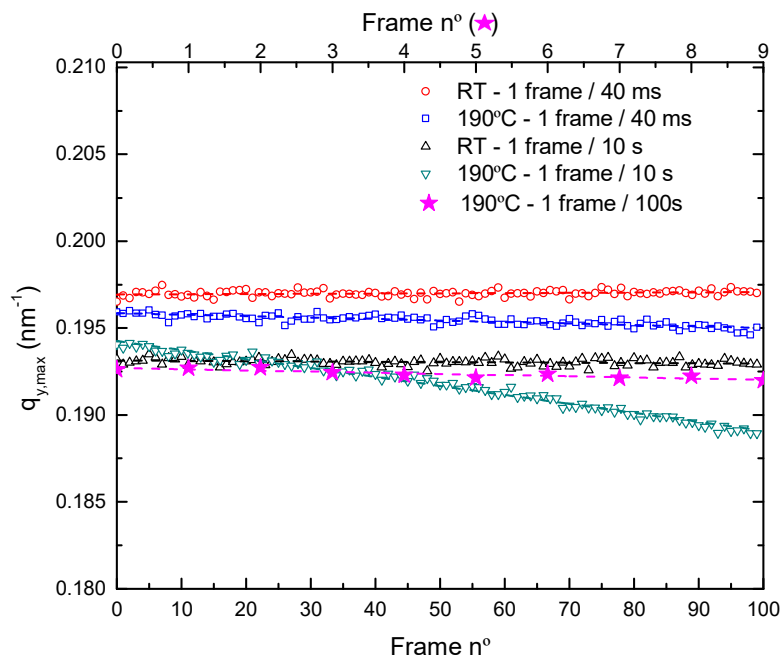
S1. Evaluation of beam damage on Block copolymer samples

Figure S1 Effect of the GISAXS beam on the block copolymer sample at room temperature (RT) and at 190°C. Evolution of Yoneda main peak position ($q_{y,max}$) at five different acquisition conditions. In the first four samples, 100 frames were taken using first 40 ms of exposure time (blue and red circles) and using 10s of exposure time (black and green triangles). As expected longer acquisition times and higher temperature caused more damage on the sample. In the last experiment, the sample was heated at 190°C and 10 frames with an acquisition time and period of 1s and 100s were taken. The annealing time and temperature of this sample is similar than previous sample but the observed beam damage was minimal.

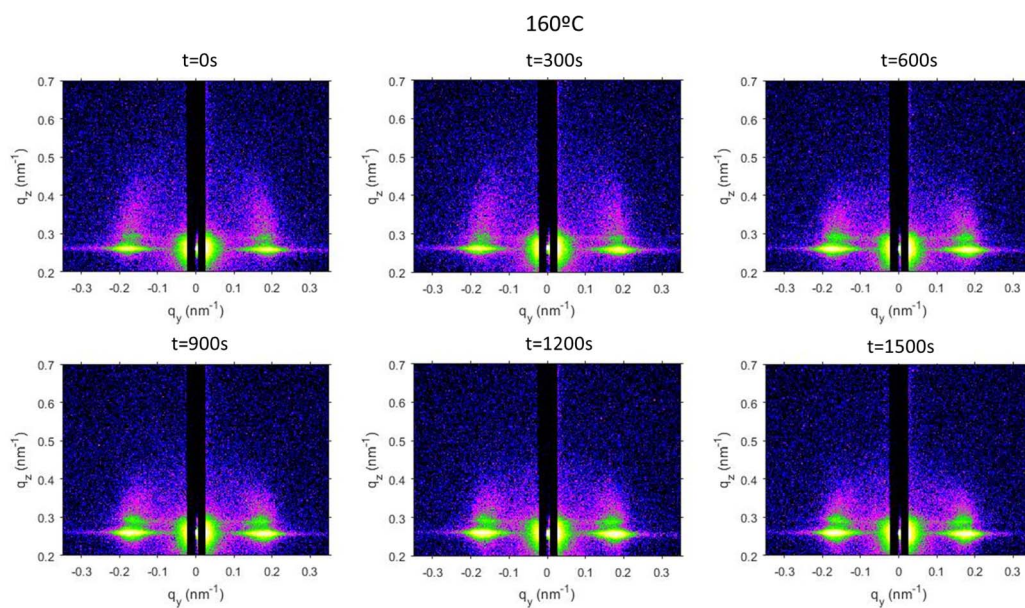
S2. Evolution of GISAXS patterns during isothermal annealing at different temperatures

Figure S2 Selection of GISAXS patterns at different processing times during the thermal annealing at 160 °C. After reaching 160 °C ($t = 0$ s) with a ramp rate of 50 °C/min, GISAXS patterns were collected with a frame rate of 1 frame of 0.037 s every 300s.

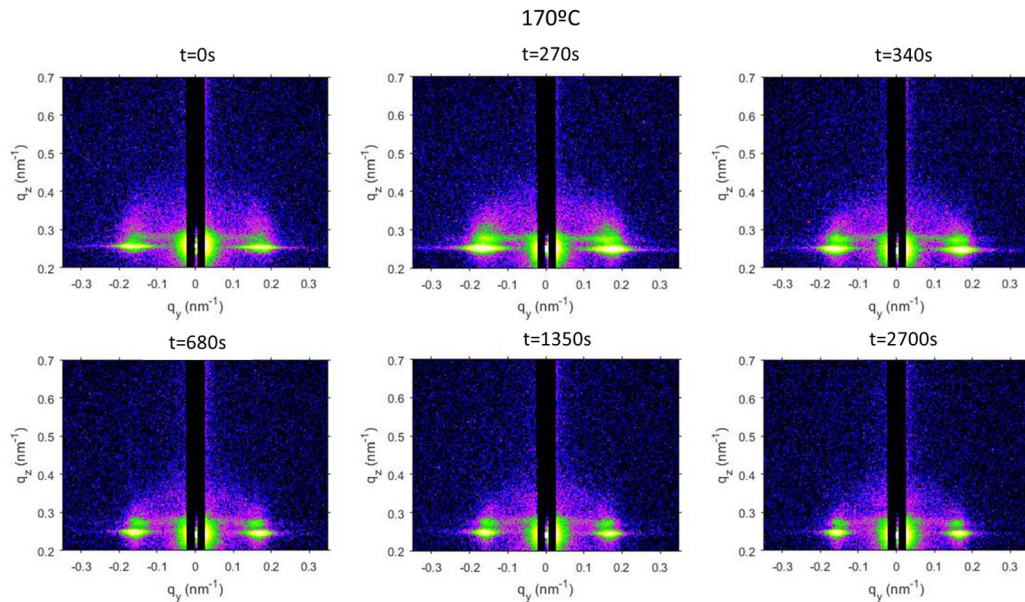


Figure S3 Selection of GISAXS patterns at different processing times during the thermal annealing at 170 °C. After reaching 170 °C ($t = 0$ s) with a ramp rate of 50 °C/min, GISAXS patterns were collected with a frame rate of 1 frame of 0.037s every 135s.

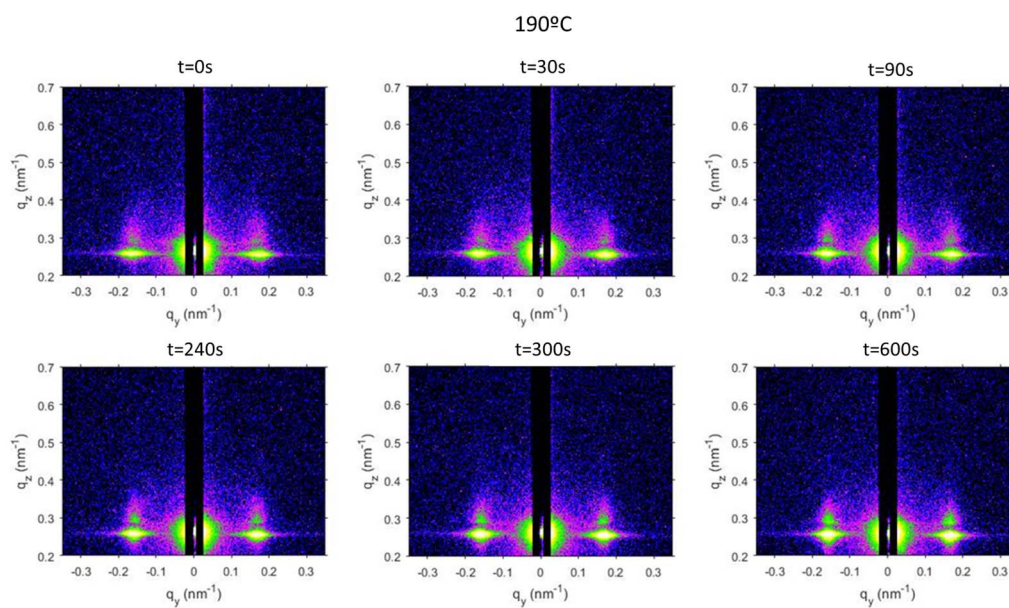


Figure S4 Selection of GISAXS patterns at different processing times during the thermal annealing at 190 °C. After reaching 190 °C ($t = 0$ s) with a ramp rate of 50 °C/min, GISAXS patterns were collected with a frame rate of 1 frame of 0.037 s every 30s.

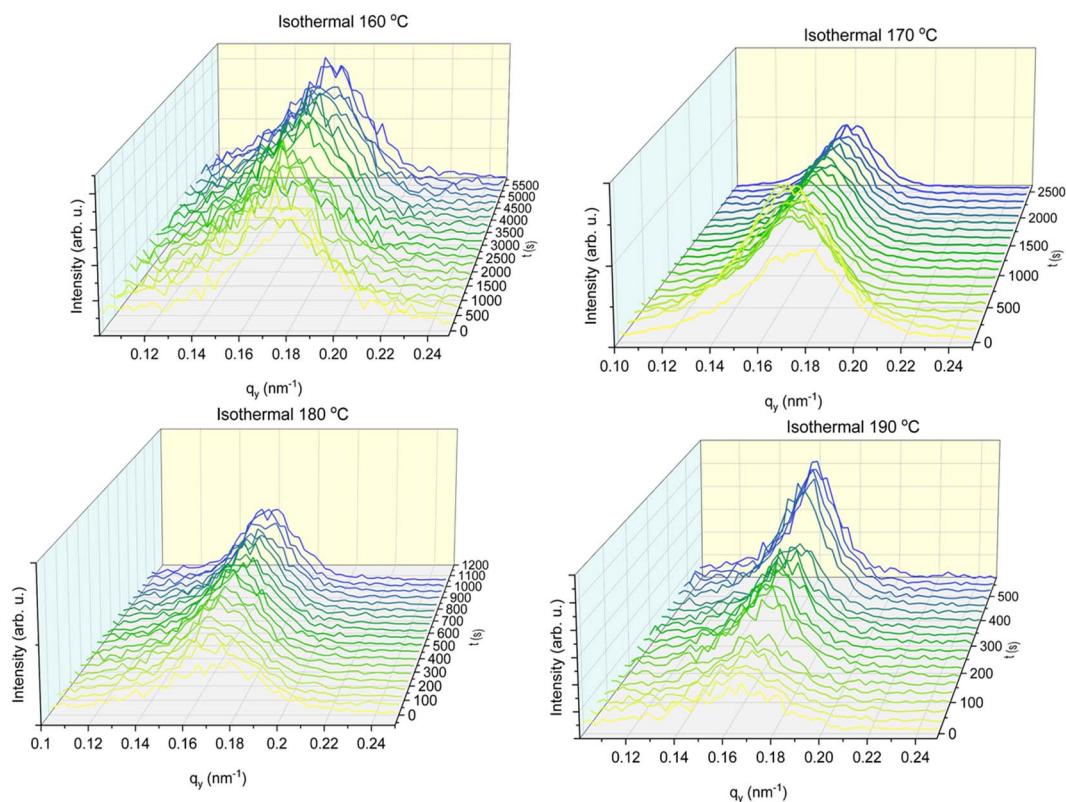


Figure S5 Detailed view of evolution with annealing time of the horizontal line cut at the first order scattering peak at Yoneda position. The profiles are extracted from the colour maps in Figure 4 (main text). The increase of the peak intensity and a reduction of the peak broadness during the thermal annealing arise from an increase of the long range order of the block copolymer pattern.

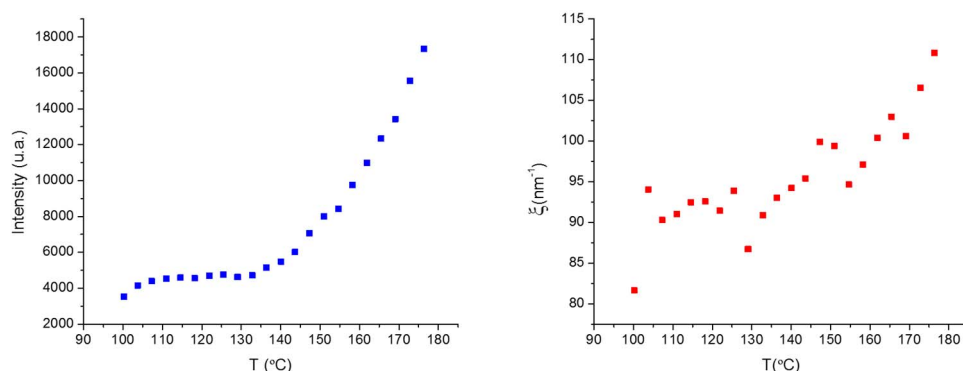
S3. Block copolymer evolution at low temperatures

Figure S6 Intensity and correlation length of the first order peak at Yoneda band plotted as a function of the temperature. The experimental values were taken from GISAXS patterns during a heating ramp at 5°C/min. At low temperatures, up to 145°C, there is not evolution on the block copolymer film, both intensity and correlation length remain constant. The initial order at lower temperatures is attributed to solvent evaporation during the spin-coating and at the beginning of the annealing process.

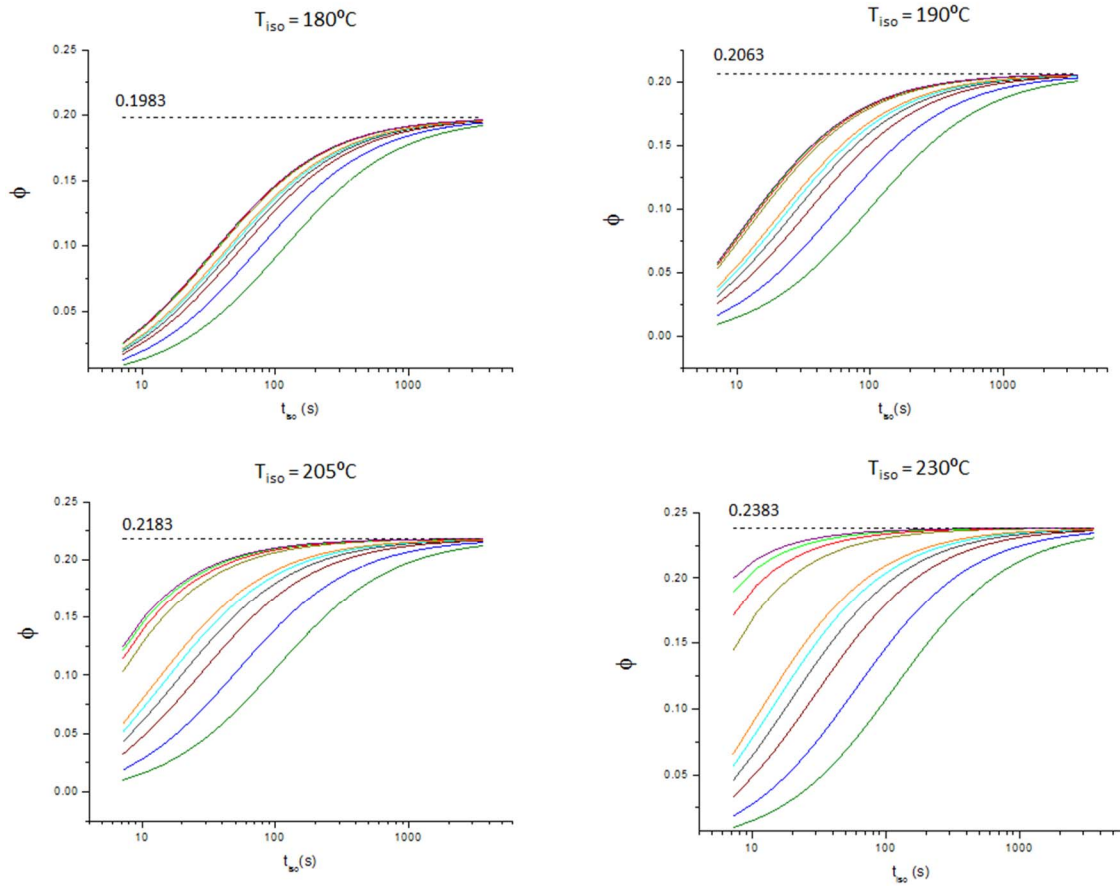
S4. Power exponent calculated using classical approach vs new model

Figure S7 Power exponent, ϕ , at different temperatures calculated from simulated curves using the classical approach, $\xi = A_T t^\phi$, from the slope of the curve $\log \xi$ vs $\log t$. The value of the power exponent calculated taking into account the equivalent time, $\xi = A_T(t + t_{eq})^\phi$, is represented by a dashed line on each graph. Before isothermal annealing, the samples are annealed from 150°C to the target temperature, T_{iso} , using a heating ramp of 1°C/s and a starting correlation length of 90nm . This demonstrates that only at long annealing times, the classical approach can be used for the calculation of the power exponent.