NATURAL AGEING OF STABILIZED AND UNUSTABILIZED LDPE FILMS USED AS GREENHOUSES COVERING MATERRIALS: ATR-FTIR AND SEM ANALYSIS

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Abstract

This article focuses on the study of both stabilized and unstabilized low-density polyethylene films aged under natural weathering in the middle of Morocco (Sidi Kacem city). The study was undertaken on three samples for each LDPE films: a new sample taken as reference; another exposed to outdoor weathering for one year; and a sample exposed for two years.

The outdoor exposure effects were followed by analysis of the total reflection Fourier transform infrared spectroscopy (ATR-FTIR). This technique provides information on the main chemical functions and evaluates the internal causes of natural ageing of our samples. The infrared spectra show the formation of hydroperoxides -O-OH, carbonyls >C=O, aliphatic esters R-(CO)-O-R, ether R-O-R, vinyl C=C, vinylidene >C=CH₂ and trans-vinylene RHC=CRH in the amorphous regions. These compounds lead to initiate the mechanism of photo-oxidation of the unstabilized LDPE samples. Therefore, the increase in the formation of oxidation products into the aged samples (LDPEUS) indicates an accelerated degradation of these latter on climatic factors (UV, O2, humidity, rain ...). The UV stabilizer (free radicals scavengers) added to the formulation of LDPE films can delay the attack of the amorphous phase and prevent the formation of compounds that can cause and accelerate photo-oxidation. The results obtained by the infrared have been confirmed by observing the morphology of the samples with scanning electron microscopy technology (SEM).

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Keywords: LDPE, Photo-oxidation, Natural ageing, UV stabilizer, ATR-FTIR, SEM

1. INTRODUCTION

Polyethylene is the most widespread polymer in the world. It belongs to the family of the polyolefin. The annual production is approximately at 80 million metric tons in 2008 [1]. Precisely, low-density polyethylene is widely used in greenhouses as covering material due to its dielectric properties combined with long durability [2], excellent chemical resistance [3, 4, 5], good resistance to cracking, low permeability to water vapor [6], low cost and easy manufacture [2, 7].

The exposure of low-density polyethylene film under natural weathering is giving rise to degradation effects. However, weathering is commonly defined as the undesirable change produced by outdoor exposure [8]. The ageing of lowdensity polyethylene results in a slow, often irreversible deterioration of the chemical properties stemming from its own instability or the effect of the environment [9]. Thus, external environment factors such as leaching by rainwater, the ultraviolet radiation (UV) and seasonal variations in temperature can cause chemical degradation of polyethylene and its weakening [10].

This work can be added to many studies have been focused on the photo-degradation of low-density polyethylene films in order to understand their mechanisms. These mechanisms can be resulted by chain-scission or crosslinking the primary structure of polymer films [11, 12]. The study of the photo-oxidation of low density polyethylene has been focused on the exposure of both commercial greenhouses to weathering conditions (outdoor) in the middle of Morocco (Sidi Kacem city). Thus, the study was undertaken on three samples for each agricultural greenhouse: a new sample taken as reference; another sample exposed to weathering for one year; and a sample exposed for two years.

The photo-degradation of low-density polyethylene film through the natural ageing was followed by testing of the attenuated total reflection Fourier transform infrared spectroscopy (ATR-FTIR). This technique provides information on the main chemical functions and evaluates the internal causes of natural ageing of our samples. Yet, the analysis by scanning electron microscopy (SEM) can reveal comparisons of the surface morphology of the samples.

2. EXPERMENTAL

2.1. Material

The commercial samples kinds (one stabilized and the other unstabilized) are supplied by the company MICATEXE (Company Production of Agricultural Films Morocco). They have densities known between 0,92 and 0,98 g.cm⁻³, thickness between 180-200 microns and weight between 165,5-176,5 g.m⁻². The average temperature of the production order is 200 °C. Two unaged samples were analyzed by ATR-FTIR spectroscopy and SEM served as reference.

2.2. Natural ageing exposure

The four kinds of samples (stabilized and unstabilized) were exposed to natural weathering. Two were exposed for one year and the other two were exposed for two years. The exposure of these samples of the low-density polyethylene under natural weathering was carried out at the roof of a building in Sidi Kacem city of Morocco.

2.3. Attenuated Total Reflection Fourier Transform

Infrared Spectroscopy

The ATR-FTIR spectroscopy is a very sensitive technique used for surface analysis. The advantage is the possibility of obtaining the spectra directly from the sheet without any further sample preparation.

The spectroscopy FT-IR is a Bruker Tensor 27 system and is installed in the laboratory spectroscopy measurements and research center of the University Ibn Tofail in Kenitra. It is equipped with a room temperature DTGS (deuterated triglycine sulfate) detector, another MCT (mercury cadmium-telluride) detector for use with the more IR-beam absorbing special samples, mid-IR source from 4000 to 400 cm⁻¹, and a KBr beam splitter. The MCT detector is about 10 times more sensitive than the DTGS detector. The sample cell in the FT-IR is a pike Miracle single-bounce attenuated total reflectance (ATR) cell equipped with a ZnSe (ZnSe absorbs strongly below 650 cm⁻¹) single crystal. This system is free to use with maximum resolution 1 cm⁻¹ [13].

The sample of low-density polyethylene is placed directly on the small crystal spot and the arm rotated over and turned down to press the sample down onto the crystal face to get better contact. The IR beam penetrates about 60 μ m into the sample. The OPUS (Optical User Software) is the Bruker data collection and analysis program for the Tensor 27 FT-IR [13].

2.4. Degree of crystallinity

The infrared spectroscopy is a technique used to determine the rate of crystallinity. The empirical equation proposed by Zerbi et al.'s used to evaluate the rate of crystallinity [14]:

$$X_{c,(\%)} = \left(1 - \frac{((I_a - I_b)/1.233)}{(I_a + I_b)}\right) \times 100$$
(1)

Where I_a and I_b are the absorbances of the peaks at 721 and 729 cm⁻¹, respectively. The constant 1.233 corresponds to the relations of the intensity bands of fully crystalline polyethylene.

2.5. Index degradation products

2.5.1. Index of vinyl

The amount of vinyl products can give the vinyl index calculation from the measurement of peak area 909 cm⁻¹ and normalized to a peak at 2912 cm⁻¹, which corresponds to asymmetric stretching vibration of CH₂ functional group that is stable to photo-degradation. The vinyl index is calculated from the following formula [15]:

$$I_{vinyl} = \mathcal{A}_{909cm^{-1}} / \mathcal{A}_{2912cm^{-1}}$$
(2)

2.5.2. Index of carbonyl

The relative absorbance of the carbonyl group is expressed by the carbonyl index. Latter is defined to describe the formation of carbonyl oxidation products [16].

$$I_{carbonyl} = \mathcal{A}_{C=0} / \mathcal{A}_{2912cm^{-1}} \tag{3}$$

Where $\mathcal{A}_{C=0}$ and $\mathcal{A}_{2912cm^{-1}}$ are the peak area of the carbonyl group was determined in the range of 1800–1670 cm⁻¹ and the reference, respectively.

2.6. Scanning Electron Microscopy

The observation of the samples in cutting or in surface was realized on a device of method ESEM with a gas pressure in the room being able to go until 26 m bar. It is also the only allowing microscope to work in three methods of different emptiness: high vacuum (10-6 mbar), low vacuum (0,1-1,33 mbar), and environmental (1,33-26 mbar). And for each of these methods a detector of the secondary electrons that furnishes non-drivers without any preparation, it also allows to observe in a dynamic manner of the materials under a given environment (pressure, temperature, gas composition). It remains topographic on the sample. The method ESEM eliminates totally the effects of loads and allows the observation of samples non-drivers without any preparation. Equipped with a complete system of microanalyses X (Detector EDX-EDAX) and of a detector of the electrons retro diffused, it allows giving the chemical composition of this [17, 18].

3. RESULTS AND DISCUSSION

3.1. Attenuated Total Reflection Fourier Transform

Infrared Spectroscopy

The infrared spectra of two new stabilized and unstabilized samples are shown in **Fig-1**. This result indicates the occurrence of the main vibration modes in the form of absorption bands that are attached to organic groups. These spectra show the presence of two regions: The first frequency is greater than 1500 cm⁻¹ and which relates to angular bending vibrations in the plane and out of plane. The second frequency is less than 1500 cm⁻¹ and which relates to stretching vibrations [19].

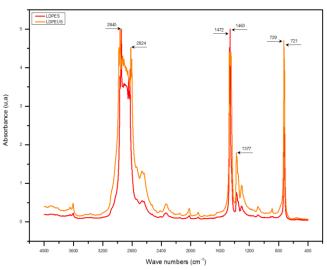


Fig-1: Infrared spectra of new LDPES and LDPEUS films

The infrared spectra show the presence of two bands with frequencies 2945 and 2824 cm⁻¹ corresponding to the stretching vibration of asymmetric and symmetric C-H bond of the methylene CH_2 group [20].

The two peaks of the doublet 1472 and 1463 cm^{-1} are attributed to the bending deformation of the methylene CH₂ groups in crystalline and amorphous phases respectively [21].

The two intense peaks as a doublet with the frequencies 729 and 721 cm⁻¹ are ascribed to the rocking deformation of the methylene CH_2 groups in crystalline and amorphous domains respectively [21, 22].

The peak at 1377cm⁻¹ with a weak intensity is assigned to CH₃ symmetric deformation crystalline [23, 24].

The spectra of the stabilized and unstabilized LDPE films aged under natural weathering for the one and two years are shown in **Figs-2** and **3**. We observe the emergence of new bands that have been considered as a starting point of the natural ageing process.

For the stabilized LDPE samples (**Fig-2**) we observe:

- Two peaks with a very weak intensity: one at 888 cm⁻¹ is assigned to vinylidene >C=CH₂ groups and the other at 831 cm⁻¹ corresponding to the vinyl R₂C=CHR groups [25, 26, 27, 28]. These unsaturations are also visible to 1635 cm⁻¹.
- Formation of the massive bands from 1200 to 1000 cm⁻¹; particularly, two peaks around 1080 and 1045 cm⁻¹ are attributed to the stretching vibration of C-O bond, this latter belongs to ether R-O-R and alcohol RCH₂-OH groups [19, 29].

For the unstabilized LDPE samples (Fig-3) we observe:

- Two extensive bands one at 3530 cm⁻¹ and the other at 3441 cm⁻¹, respectively assigned to free and associated hydro-peroxides (O-O-H) groups [21, 28, 30].

- A broad band at 3369 cm⁻¹ assigned to all of the hydroxyl (O-H) groups [28].
- Two peaks of high intensity at 1712 and 1733 cm⁻¹ were attributed to C=O stretching vibration of ketone and aliphatic esters respectively [31, 32]; ester group presents additional C-O bond confirmed by the presence of the peak at 1180 cm⁻¹[31, 32].
- Important massive bands between 1000 and 1200 cm⁻¹; particularly, two peaks 1083 and 1180 cm⁻¹ are attributed to stretching vibrations of the C-O, and belong to ether and aliphatic esters groups respectively [19, 29, 33].
- Formation of the massive bands between 850 and 1000 cm⁻¹; specifically, Two peaks 994 and 909 cm⁻¹ corresponding to the vinyl RCH=CH₂ groups, another peak rises around 962 cm⁻¹ attributed to trans-vinylene RHC =CRH groups and finally, one peak at 888 cm⁻¹ attributed to vinylidene >C=CH₂ groups [25, 26, 27, 28]. These unsaturations are also visible to 1641 cm⁻¹.

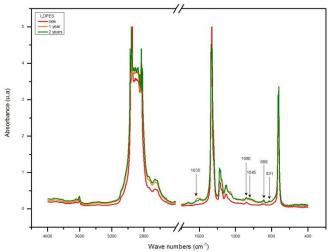


Fig-2: Infrared spectra of LDPES samples after exposure to natural weathering

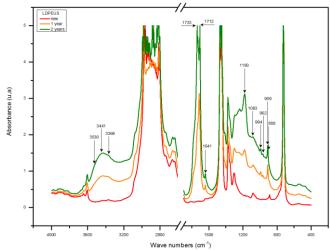
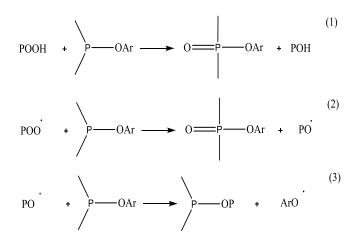


Fig-3: Infrared spectra of LDPEUS samples after exposure to natural weathering

The absence of the bands attributed to hydroperoxides groups in the infrared spectra of stabilized samples (Fig-2)

can relate to the presence of hydroperoxides decomposers associated with phenolic antioxidants. The compounds most commonly used to decompose hydroperoxides are trivalent phosphites. We also decompose hydroperoxides (**react 1**); the phosphites could appear as reducing peroxyl (**react 2**) and alkoxyl (**react 3**) radicals [33, 34].



However, phenolic antioxidants (Butylated hydroxytoluene (BHT)) are most widely used to capture the free radicals [33]. The reaction equation (**react 4**) is given as follows [34, 35]. According to [33], the phenoxy radicals formed are stabilized by isomerism.

$$AH + POO \longrightarrow \dot{A} + POOH$$
 (4)

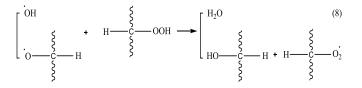
The appearance of bands attributed to hydroperoxides (**Fig. 3**) results in the absence of decomposers of hydroperoxides associated with phenolic antioxidants. The preferential oxidation of the amorphous phase relative to the crystalline phase is a result of the higher diffusion of oxygen within it. The hydroperoxides are extremely photo-labile. They absorb UV radiation and initiate photo-oxidation [36, 37]. This priming can cause weakness of the initial split O-O bond (E_d = 176 kJ mol⁻¹ [38]) in hydroperoxides (**reacts 5-7**) [39, 40].

ROOH
$$\xrightarrow{hv}$$
 \dot{RO} + OH \xrightarrow{RH} \dot{R} + H_2O/ROH (5)

$$\dot{R} + O_2 \longrightarrow RO_2^{-}$$
 (6)

$$RO_2 + RH \longrightarrow ROOH + \dot{R}$$
 (7)

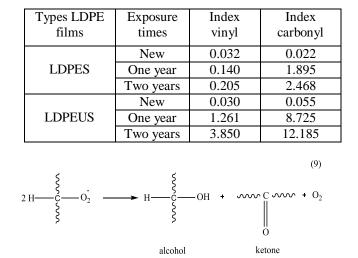
The free-radical products can then attack adjacent sechydroperoxides sites to generate new sec-peroxyl radicals [28] (**react 8**).



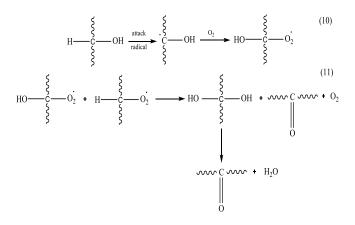
The results presented in **Table-1** show a significant increase in the index of carbonyl compounds for unstabilized greenhouses. This increase results from the self-reactivity of peroxyl radicals (**react 9**) produced from the split of hydroperoxides in the presence of oxygen [41, 28] (**react 6**). Therefore, hydroperoxides are precursors of carbonyl species (ketone) and alcohols.

 Table-1: Index vinyl and carbonyl of the different samples

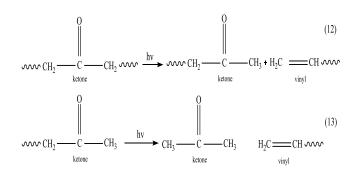
 LDPES and LDPEUS



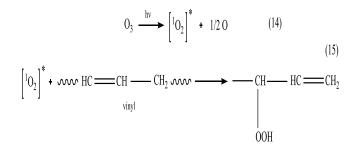
Therefore, the alcohol produced in the above reaction can be attacked to give an alkoxyl [37, 42] (**react 10**); then it can be transformed into alko-pyroxyl in the presence of atmospheric oxygen. The alkoxyl can react with another peroxyl radical to give a ketone [28] (**react 11**).



The increase in the index of vinyl compounds is explained by the photolysis of ketones according to the mechanism of Norrish II. This mechanism (photo-cleavage of ketones) gives vinyl compounds (**reacts 12, 13**) [28]. Norrish type I cleavage will also destroy the ketone groups but is much slower than the type II process for backbone ketones [43, 44]. The photolysis of ketones according to the Norrish I give alkyl radicals.



In addition to the photolysis of hydroperoxides and carbonyl compounds, other priming is possible of the photo-oxidation resulting from transfer of the charges complexes caused by the presence of catalytic residues or also ozone. Indeed, ozone photolysis by UV radiation (**react 14**) leads to very reactive species such as singlet oxygen ($^{1}O_{2}$). It reacts directly with double bonds and leads to the rupture of chains (**react 15**) [45].



Therefore, for LDPEUS we described the increase of 7.26% the degree of crystallinity after two years (Table.2).This increase due to the molecular chain scission according to the mechanism of Norrish II. Thus, the photo-oxidation makes the amorphous part freer and mobile for recrystallization [46]. However, for LDPES we have a small increase of 1.74% the degree of crystallinity due to the molecular chain scission according to the Norrish I greater than the Norrish II.

 Table-2: Crystallinity of the different samples LDPES and LDPEUS

Types LDEP	Exposure	Degree of crystallinity
films	times	%
LDPES	New	61.60
	One year	62.70
	Two years	63.34
LDPEUS	New	65.70
	One year	69.47
	Two years	72.96

The formation of hydroperoxides O-OH, carbonyl C=O, vinyl C=C and trans-vinylene RHC=CRH groups (band at 962 cm⁻¹) leads to initiate the mechanism of photodegradation of the LDPEUS in a way more advanced than the LDPES. However, the UV stabilizer added to the formulation of LDPE can delay the attack of the amorphous phase and prevent the formation of compounds (hydroperoxides and carbonyls) that can cause and accelerate photo-oxidation.

3.2. Scanning Electron Microscopy

The micrographs of scanning electron microscopy (SEM) of two low-density polyethylenes are shown in Fig- 4-9. These show a difference between the morphology of the samples during the natural weathering exposure. We observe the presence of some white spots due to the existence of some vacuum from the liberation of gas during the treatment. Yet, the micrographs can show that the surface of the aged unstabilized LDPE samples is attacked more than the stabilized LDPE samples. This is explained by the increase in the amount of absorbed ultraviolet radiation and increased oxygen diffusion in the samples. Thus, the change in temperature, humidity and rainfall are climatic parameters capable of altering the surface morphology of the samples. These morphologies are a consequence of the increase in photo-oxidation reactions that brings breaking of chemical bonds of the macromolecular backbone of polymer.

Moreover, the existence of dust and air pollutants can affect the mechanism of degradation of the polymer surface. These results show that the unstabilized LDPE films undergo a remarkable degradation of climatic factors (UV, O₂, humidity ...), which confirms the results obtained by infrared spectroscopy.

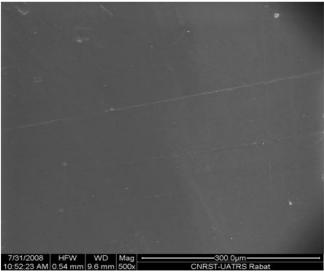


Fig-4: SEM micrograph of virgin LDPES sample

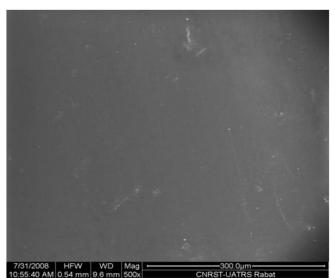


Fig-5: SEM micrograph of LDPES sample after one year

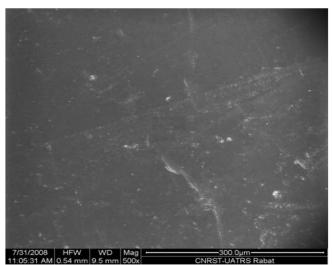


Fig-6: SEM micrograph of LDPES sample after two years

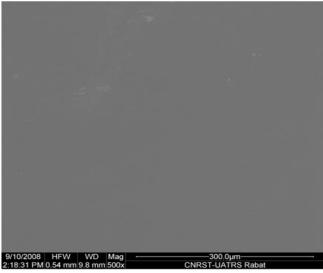


Fig-7: SEM micrograph of virgin LDPEUS sample

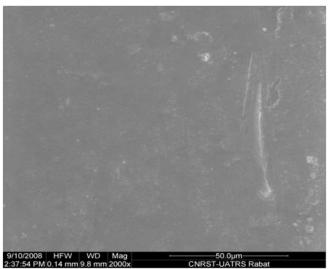


Fig-8: SEM micrograph of LDPEUS sample after one year

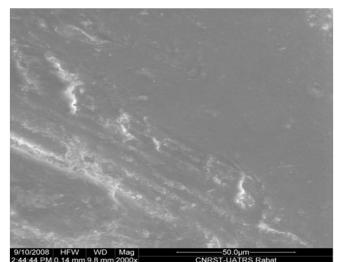


Fig-9: SEM micrograph of LDPEUS sample two years

4. CONCLUSION

The ATR-FTIR and scanning electron microscopy (SEM) were used to study the natural ageing of low-density polyethylene greenhouses covering films.

The study by infrared analysis allowed comparisons between the spectra of three samples for each commercial greenhouses (stabilized and unstabilized) to identify different chemical groups of low density polyethylene. This analysis shows the formation of hydroperoxides -O-OH, carbonyls >C=O, aliphatic esters R-(CO)-O-R, ether R-O-R, vinyl C=C, vinylidene >C=CH₂ and trans-vinylene RHC=CRH (962 cm⁻¹) in the amorphous regions. These compounds lead to initiate the mechanism of photooxidation of the unstabilized LDPE samples. The preferential oxidation of the amorphous phase relative to the crystalline phase is a result of the higher diffusion of oxygen within it. The peaks around this latter phase are due to oxidation processes. Photo-oxidation is generally considered to be a result of oxidative processes in the amorphous regions, which change the primary structure of a polymer by chain-scission or crosslinking [47, 48]. The increase in the intensity of the carbonyl >C=O band (1712 cm⁻¹) due to the accumulation and decomposition of hydroperoxides compounds were formed during outdoor exposure. Thus, the increase in the intensity of the vinyl compounds is explained by the photolysis of ketones according to the mechanism of Norrish type II.

Absorption of a photon of total solar radiation can activate a particular bond or group in a macromolecule. The most deleterious part of total solar radiation is the ultraviolet (UV) radiation (range between 290 and 400 nm) which may be the most important factor in the degradation of polymers used for outdoor purposes [49]. The UV stabilizer (free radicals scavengers) added to the formulation of LDPE films can delay the attack of the amorphous phase and prevent the formation of compounds that can cause and accelerate photo-oxidation.

Therefore, the increase in the formation of oxidation products into the aged samples (LDPEUS) indicates an accelerated degradation of these latter on climatic factors (UV, O2, humidity, rain ...). The results obtained by the infrared have been confirmed by observing the morphology of the samples with scanning electron microscopy technology.

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